

Incidence of the mechanisms of damage in the scale effect of composites involving Ultra Thin Plies

Federico París

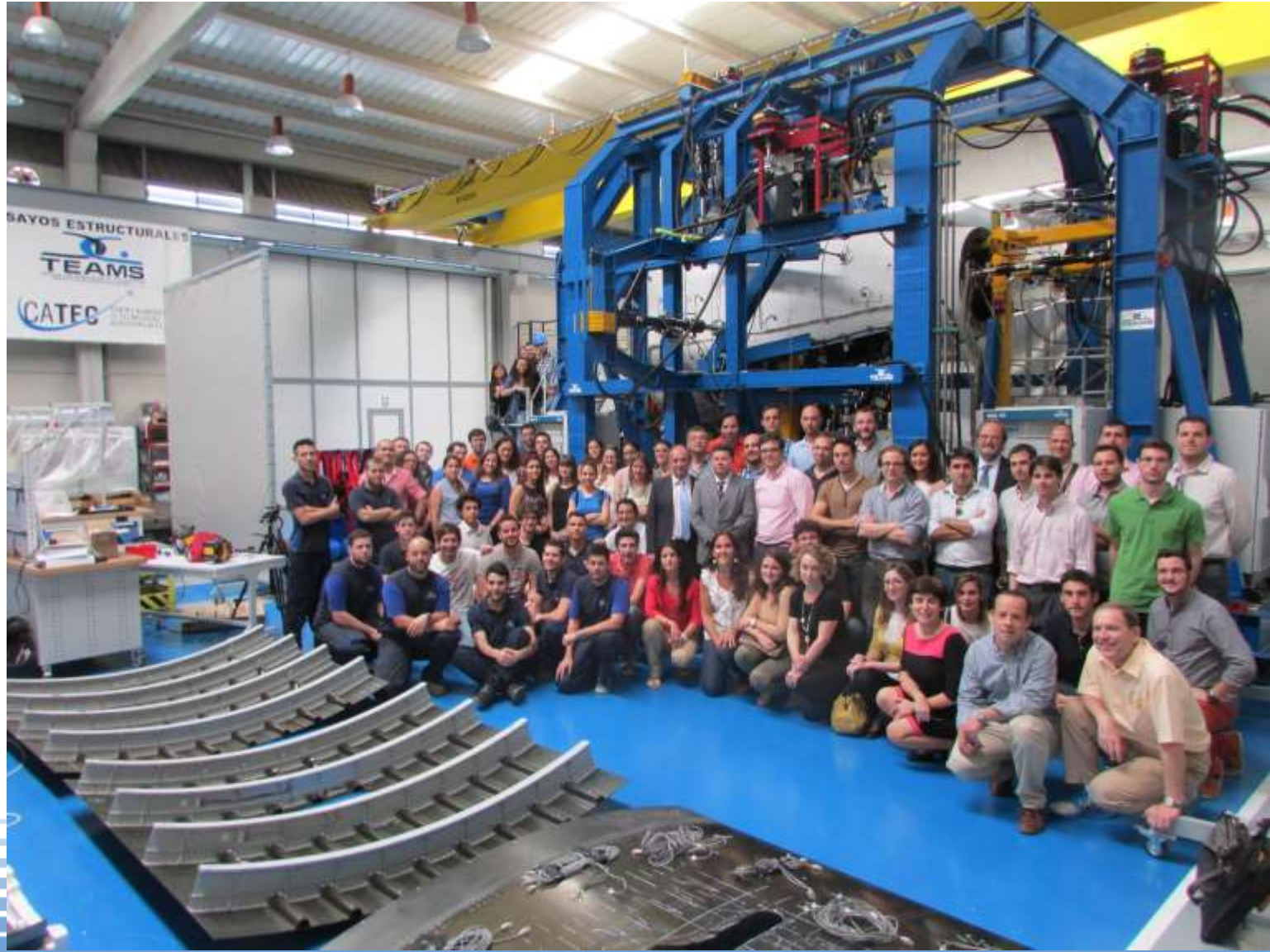
Group of Elasticity and Strength of Materials
Department of Continuum Mechanics
School of Engineering, University of Seville, Spain

COMPTEST

Girona, May 30-June 2, 2023

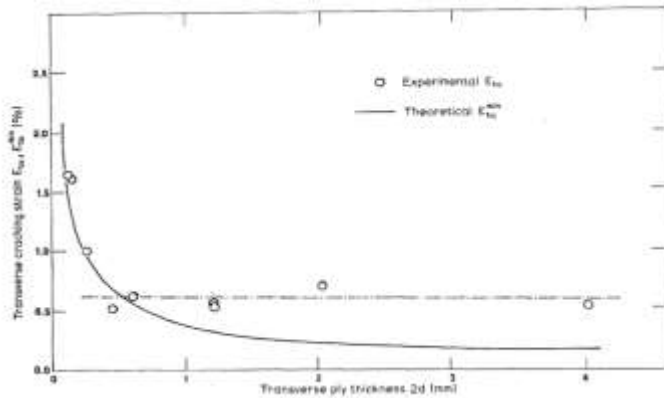
THE BACKGROUND

FULL SCALE TEST OF TAIL CONE OF THE A350



Parvizi et al
(1978)

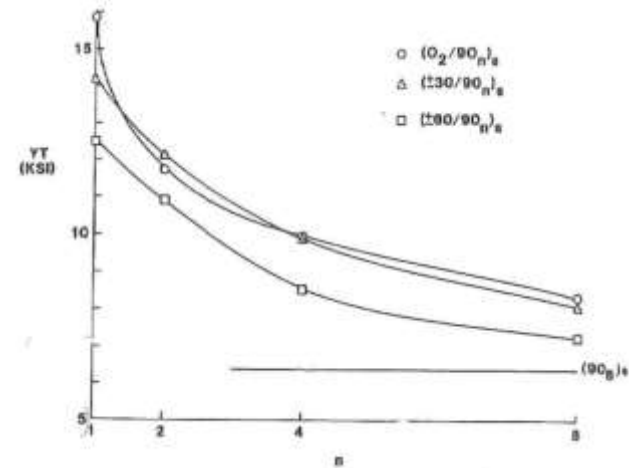
Glass fibre /epoxy resin
Detection Method: visual and
acoustic emission



Values of ϵ_{tu} as a function of ply thickness and
experimental values of ϵ_{tu} for various ply thicknesses

Flaggs and Kural
(1981)

Carbon fibre /epoxy resin
Detection Method: x-radiographic
inspection



In situ transverse strengths as a function of
thickness and the orientations of the
adjacent laminae.

The damages that appear were not specified, it was understood that
the points in the graphics were associated with the appearance of a
transverse crack in the 90 degrees lamina

NASA/CR-2001-210661



A Study of Failure Criteria of Fibrous Composite Materials

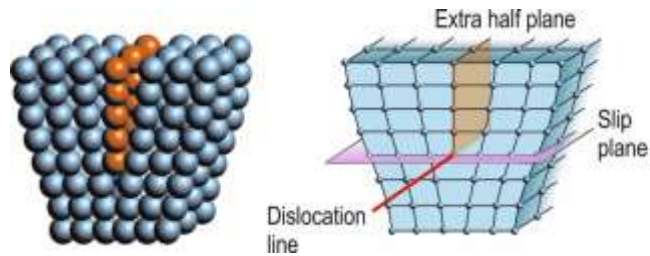
*Federico Paris
George Washington University
Joint Institute for the Advancement of Flight Sciences
Langley Research Center
Hampton, Virginia*

Damage: Alteration of the internal structure of the material abandoning the pristine state.

What is the key factor to predict damage?

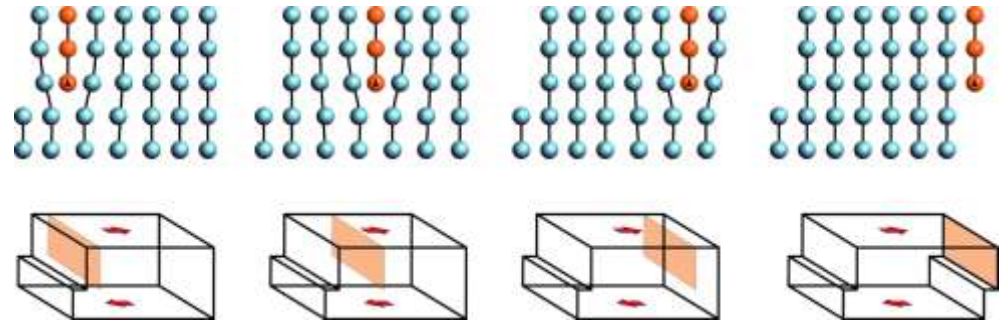
Damage usually takes place at micromechanical level and typically we take homogenized models where the variables involved (stresses or strains for instance) are associated with homogenized (macromechanical) models of the material.

Then, in order to get success in representing the damage by means of variables associated with the homogenized material, we should have a physical connection between the damage at micromechanical level and the variable used to predict the damage at macromechanical level.



Schematic representation of a dislocation

Movements of a dislocation:
final configuration of the material



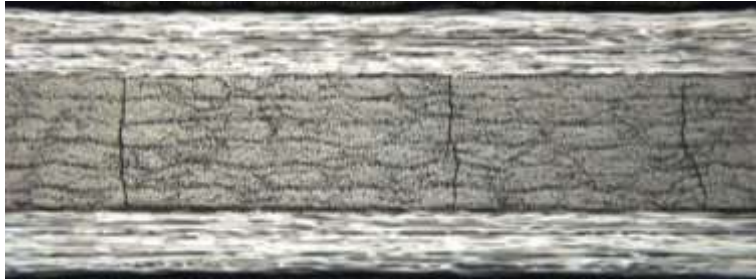
Von Mises criterion:

$$|S^d| < \sqrt{2/3} S_e$$

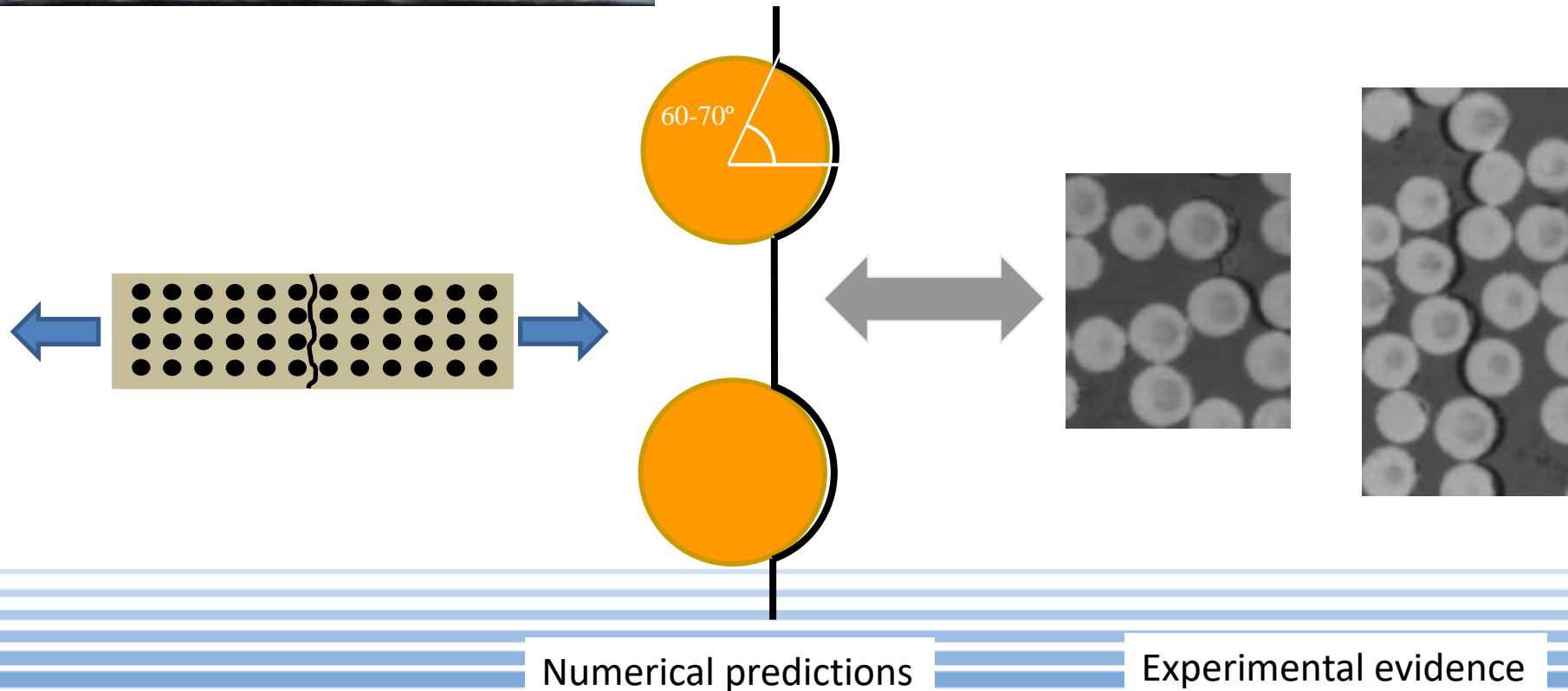
Hencky version:

$$U(S_{ij}^d) < S_e^2 / 6G$$

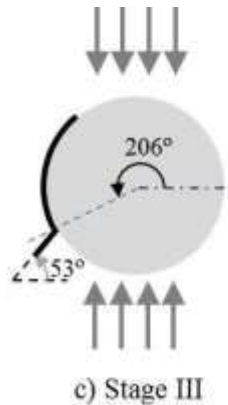
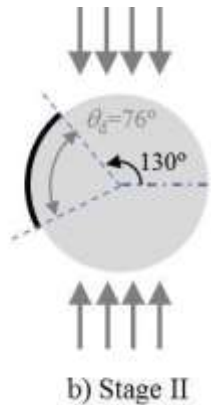
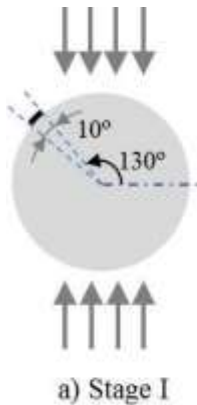
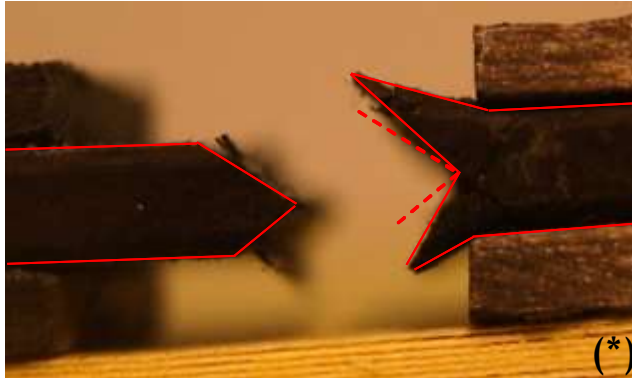
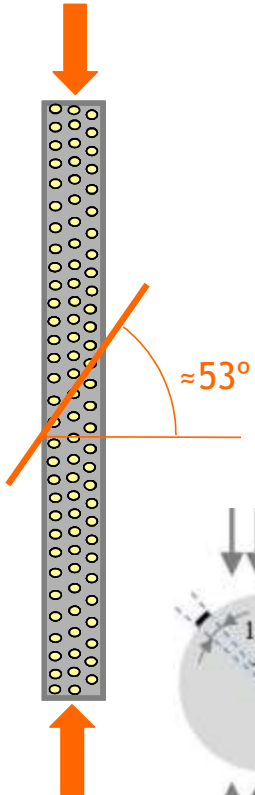
THE TOOL: MICROMECHANICAL STUDY OF TRANSVERSE FAILURE UNDER TENSION



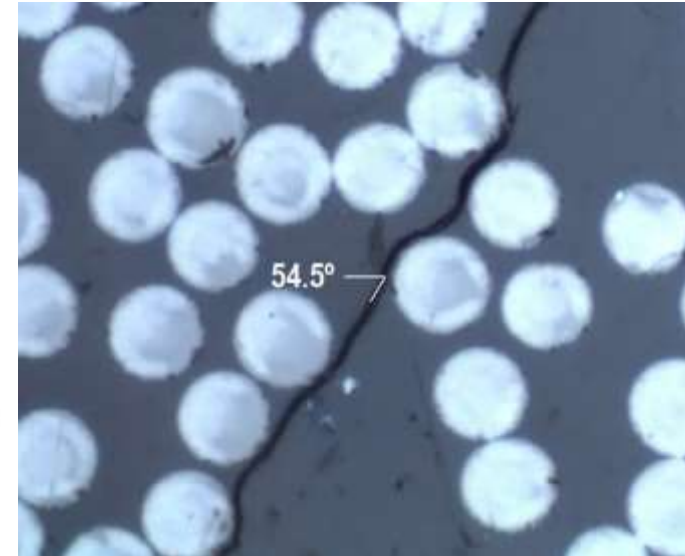
París, F., Correa, E., Mantič, V. Kinking of transverse interface cracks between fibre and matrix. *Journal of Applied Mechanics-Transactions of the ASME* 74:703-716, 2007.



THE TOOL: MICROMECHANICAL STUDY OF TRANSVERSE FAILURE UNDER COMPRESION

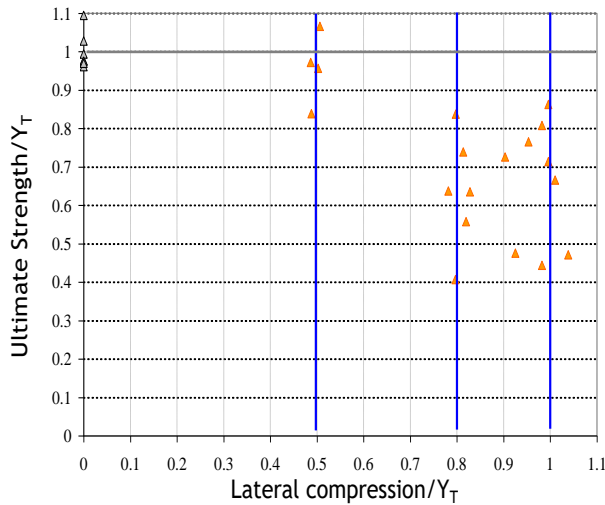
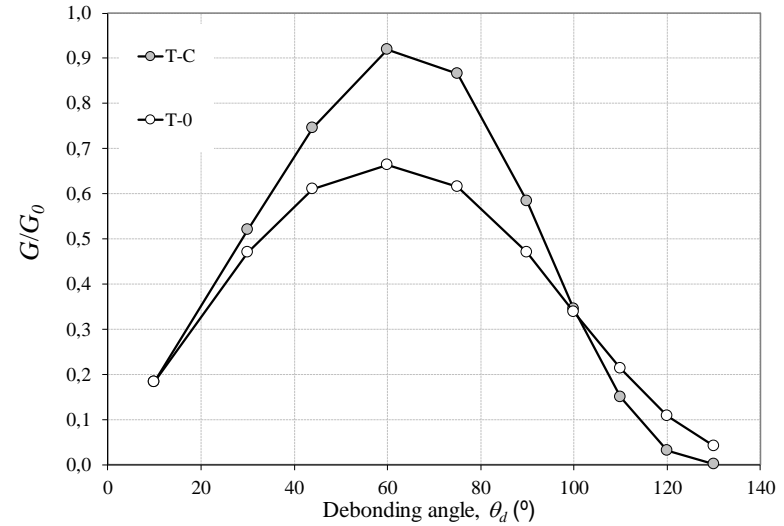
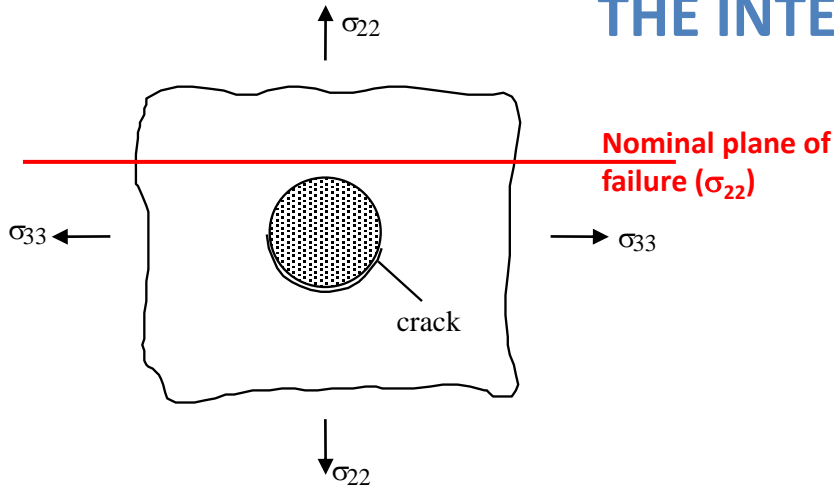


Experimental evidence



Correa E., Mantič V., París F., A micromechanical view of inter-fibre failure of composite materials under compression transverse to the fibres, *Comp. Sci. Tech.*, 68, pp. 2010-2021, 2008.

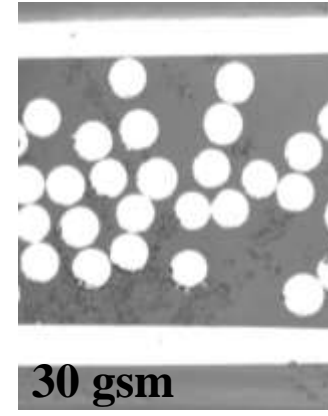
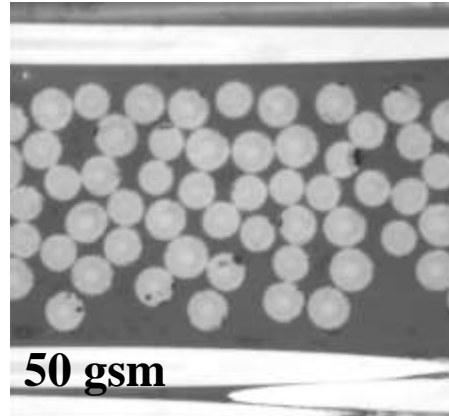
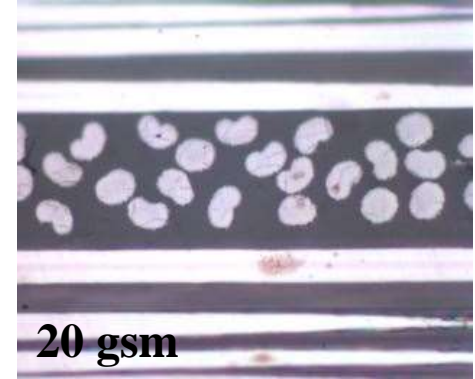
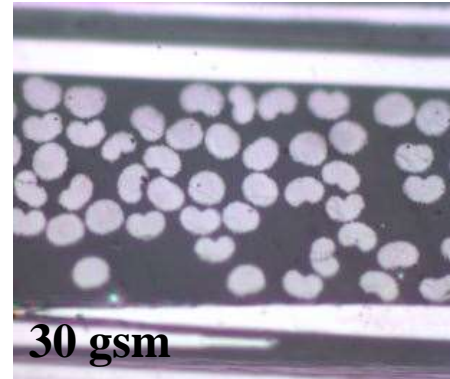
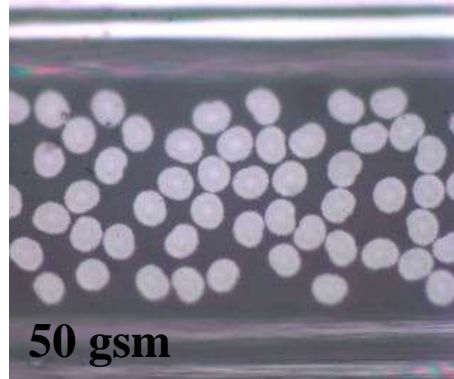
THE TOOL: MICROMECHANIC EFFECT OF THE STRESS PARALLEL TO THE PLANE OF FAILURE IN THE INTERFIBRE FAILURE

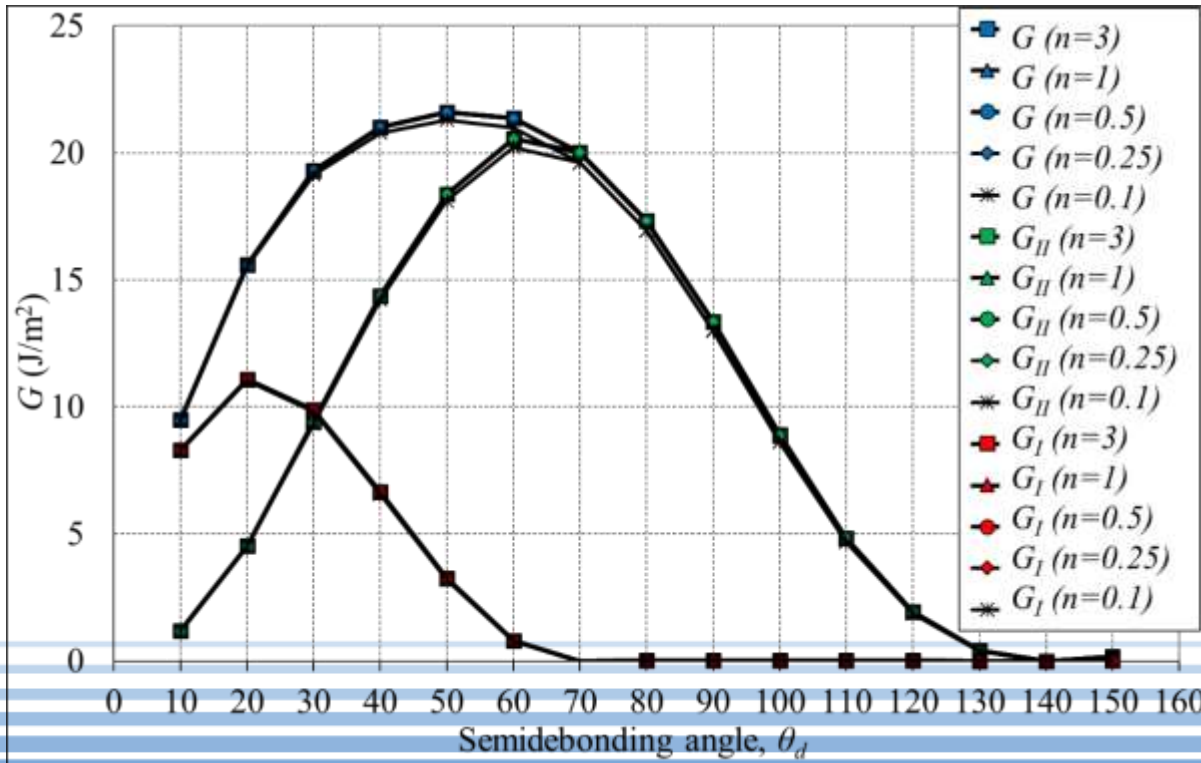
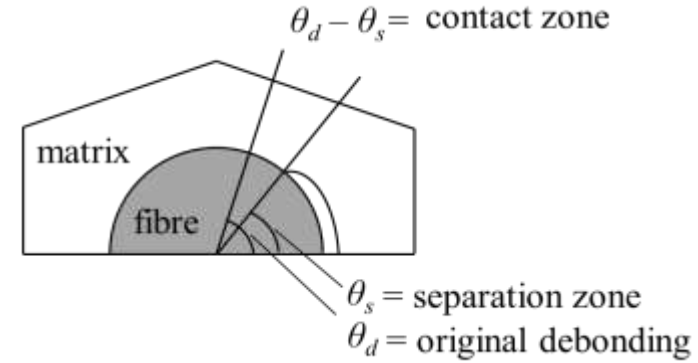
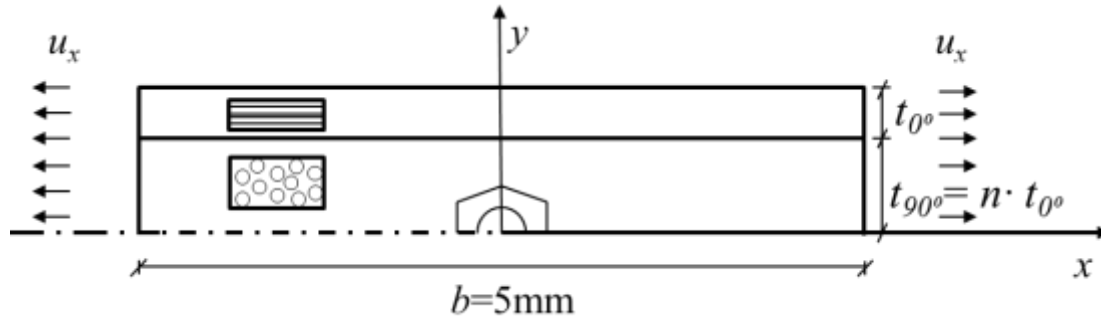


The presence of a lateral compression superimposed on the tension nominally responsible for the failure accelerates the fracture

E. Correa, F. París, V. Mantič, Effect of the presence of a secondary transverse load on the inter-fibre failure under tension. *Engineering Fracture Mechanics* 103: 174-189, 2013.

TRIPLE H COMPOSITES
UK SUPPLIER OF
SKYFLEX Carbon Prepreg



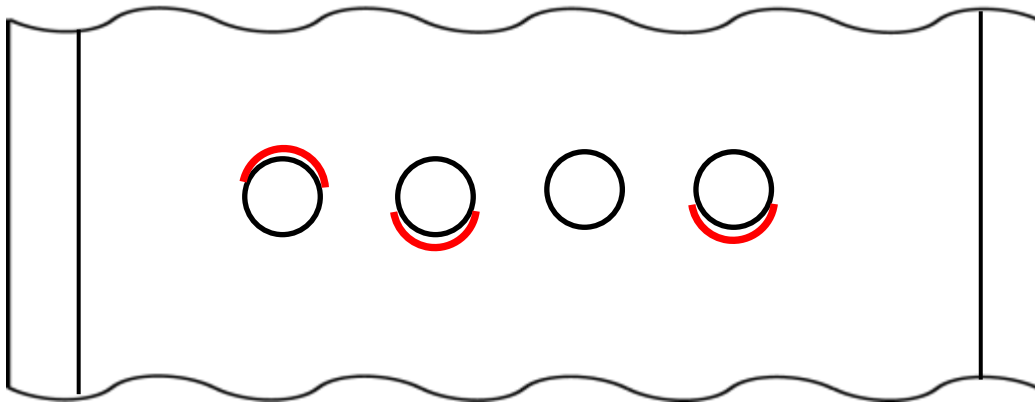


There is the same probability of getting isolated debondings in laminates with thick and thin 90 degrees laminas.

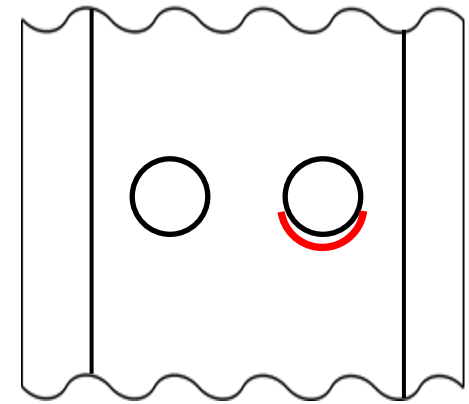
Same probability of getting isolated debondings in laminates
with thick and thin 90 degrees laminas,

but it does not imply the same possibility of observing them!

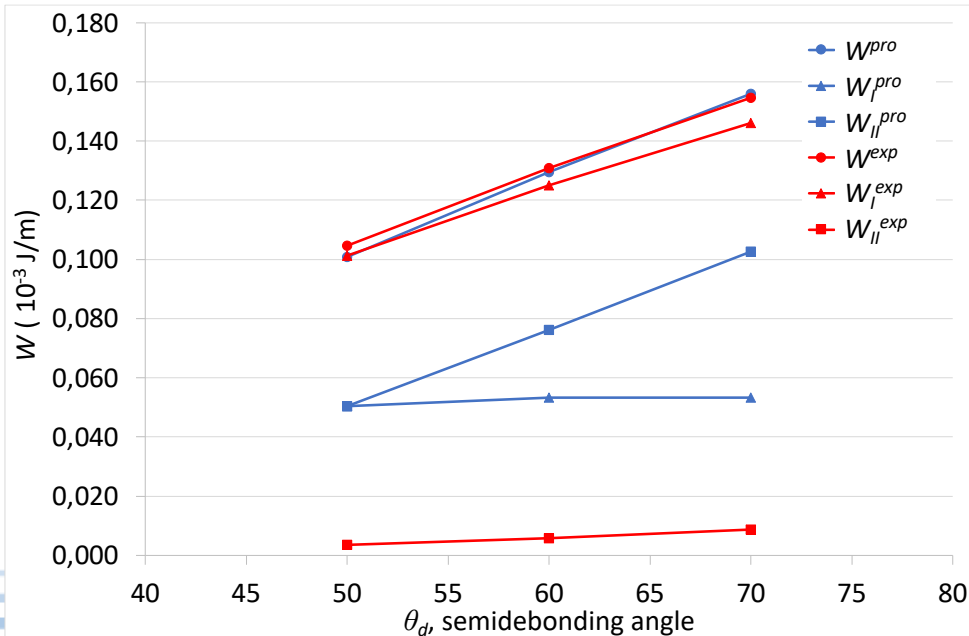
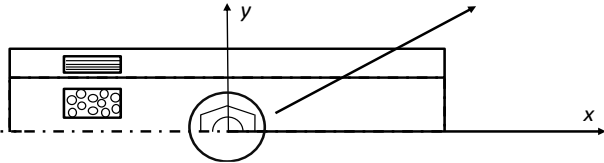
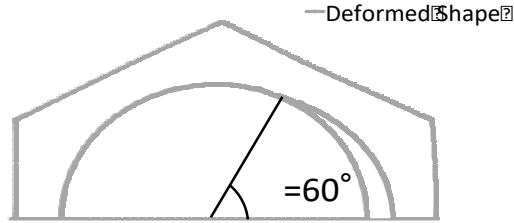
In fact, we had never observed isolated debondings, as a first manifestation of
damage using plies of conventional thickness.



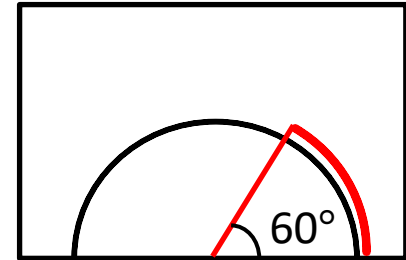
Thick (90°) laminates



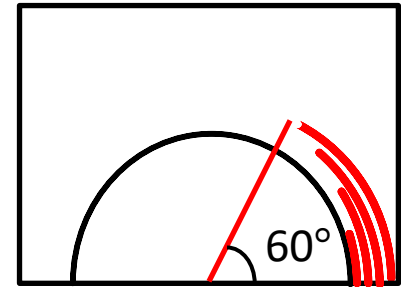
Thin (90°) laminates



Explosive Mechanism

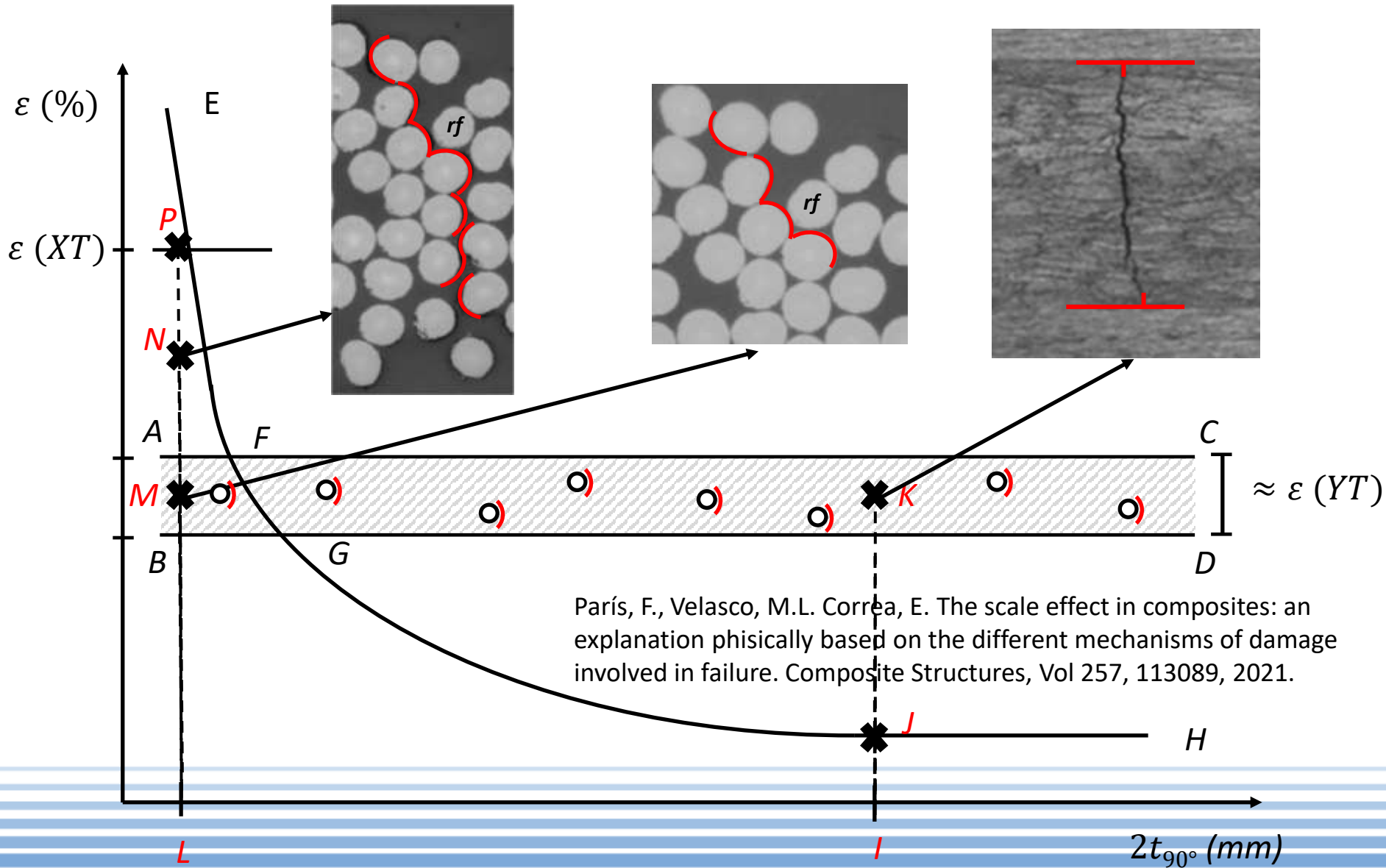


Progressive Mechanism



Isolated debonding between fibre and matrix which appears in an explosive or progressive way.

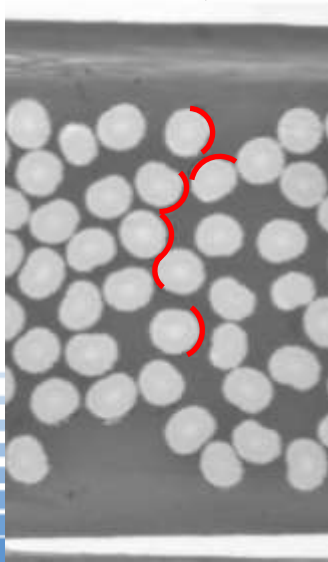
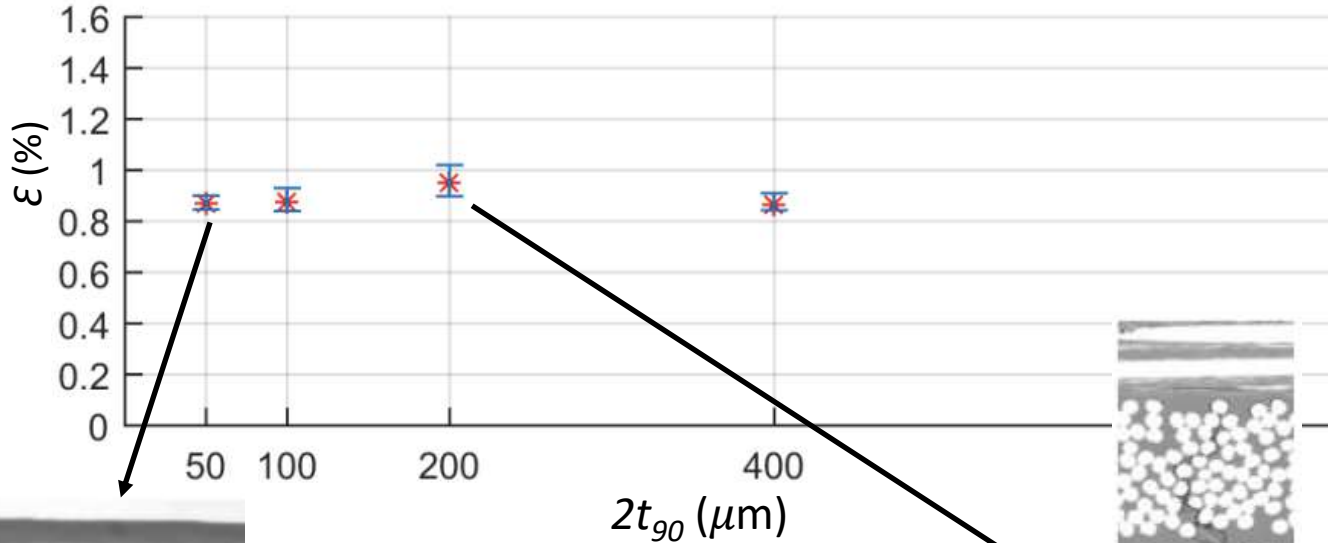
A micromechanical analysis using FEM with a cohesive approach, for instance, may condition the detection of one particular mechanism of damage



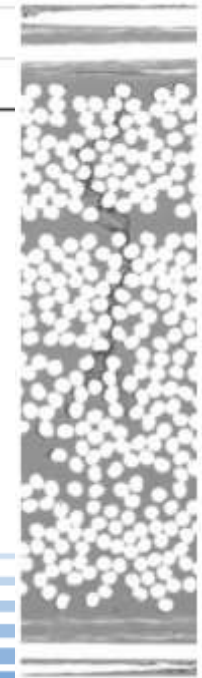
París, F., Velasco, M.L. Correa, E. The scale effect in composites: an explanation physically based on the different mechanisms of damage involved in failure. *Composite Structures*, Vol 257, 113089, 2021.

Detection of the FIRST damage

$[0_{12}, 90_n, 0_{12}]$ LAMINATES MADE OF UTPS OF USN50



**No possibility of checking it
till the appearance of UTPs**



$[0_{12}, 90, 0_{12}]$
 $2t_{90}=50\text{gsm}$

$[0_{12}, 90, 0_{12}]$
 $2t_{90}=200\text{gsm}$

PRACTICAL APPLICATION OF THE KNOWLEDGE ACQUIRED



-Verification of the pattern of damage associated with the failure of the 90 degrees lamina as a function of the thickness

PRACTICAL APPLICATION OF THE KNOWLEDGE ACQUIRED

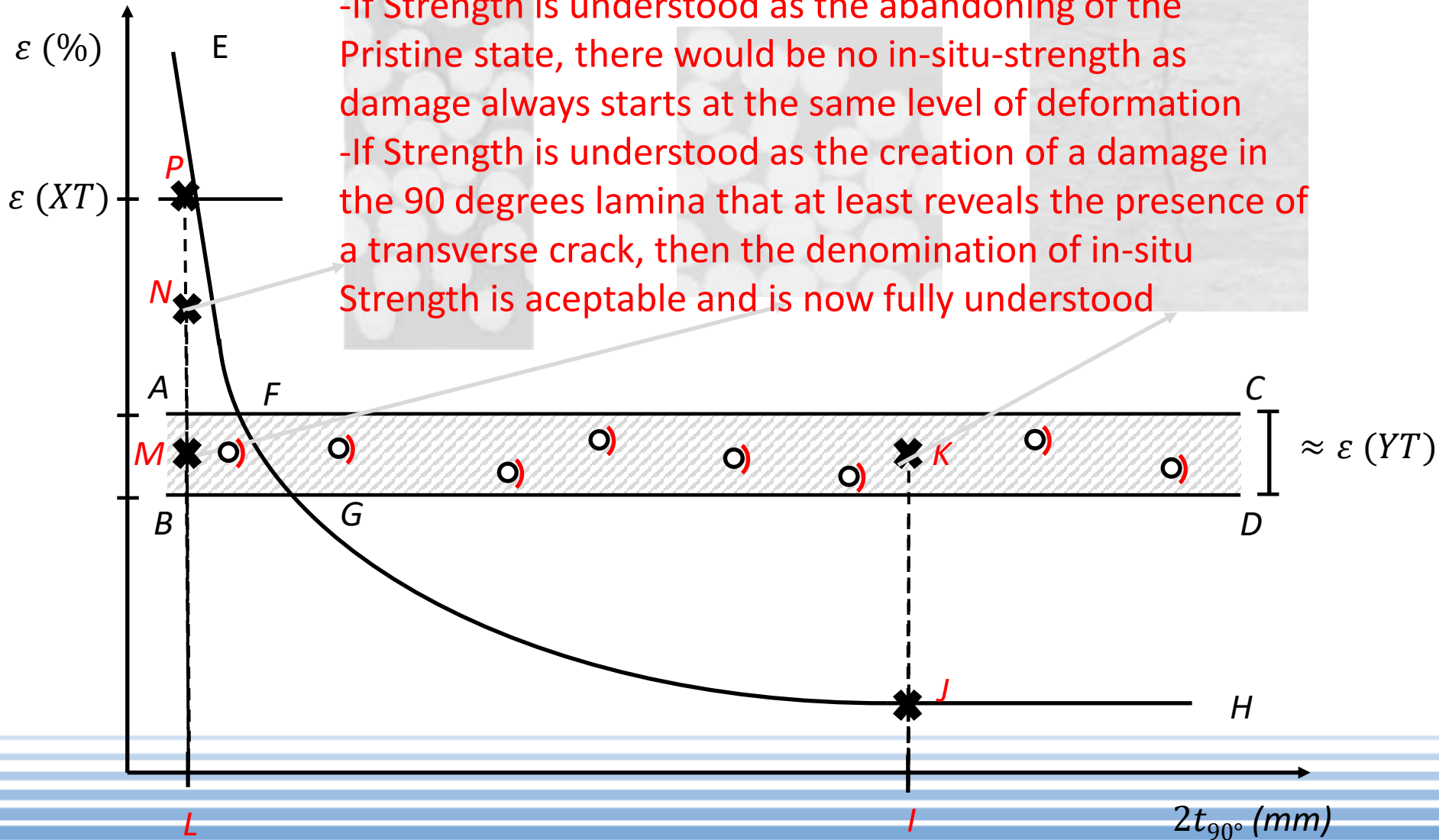


-Verification of the pattern of damage associated with the failure of the 90 degrees lamina as a function of the thickness

-On the concept of “in-situ Strength”

What do we understand by STRENGTH?

- If Strength is understood as the abandoning of the Pristine state, there would be no in-situ-strength as damage always starts at the same level of deformation
- If Strength is understood as the creation of a damage in the 90 degrees lamina that at least reveals the presence of a transverse crack, then the denomination of in-situ Strength is acceptable and is now fully understood

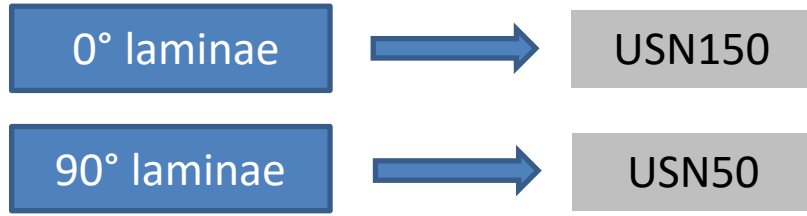


PRACTICAL APPLICATION OF THE KNOWLEDGE ACQUIRED

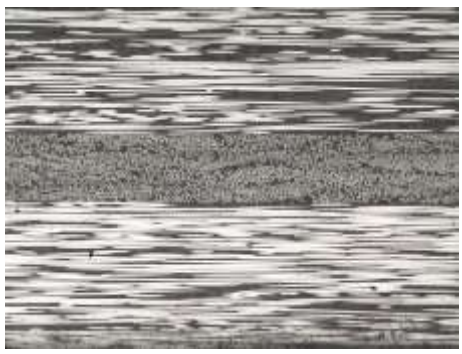


- Verification of the pattern of damage associated with the failure of the 90 degrees lamina as a function of the thickness
- On the concept of “in-situ Strength”
- Optimization of the thickness of the 90 degrees laminas

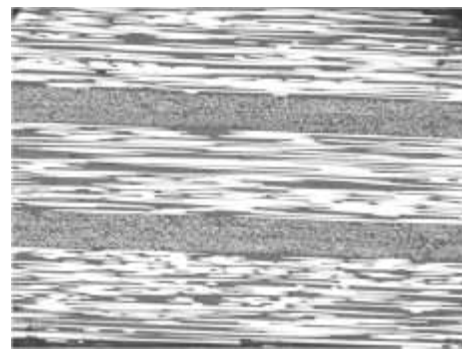
OPTIMIZATION OF THE MATERIAL



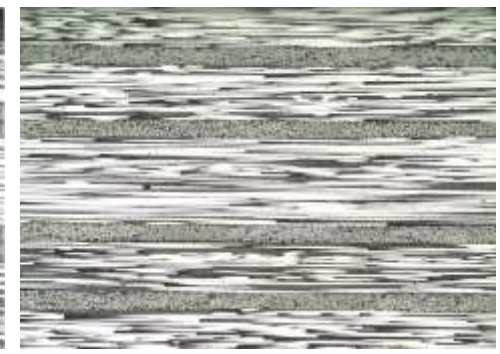
Total thickness of the laminates = 1,1mm



[0₃, 90₄, 0₃]



[0₂, 90₂, 0₂, 90₂, 0₂]

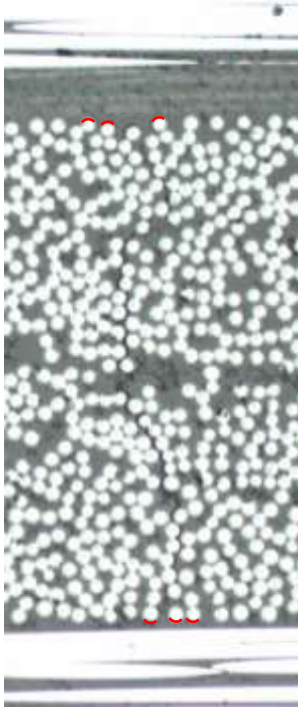


[0, 90, 0, 90, 0, 0, 90, 0, 90, 0]

PROGRESSION OF DAMAGE IN LAMINAS ORIENTED 90°

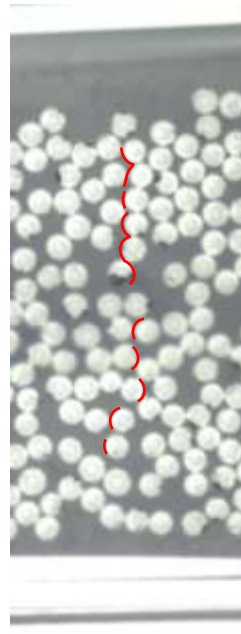
12000N-13000N
 $\epsilon \sim 0.5\%$

[0₃, 90₄, 0₃]



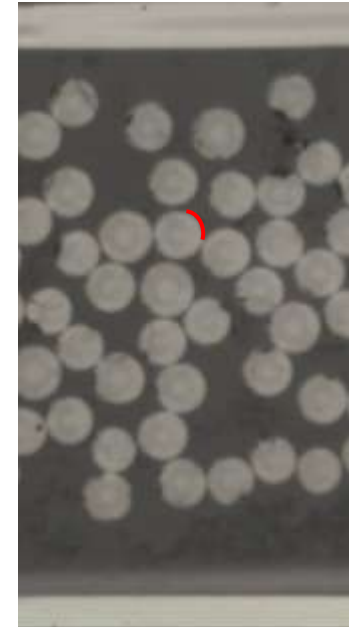
TRANSVERSE DAMAGE WITH
ADDITIONAL DAMAGE INDICATING
FUTURE DELAMINATION

[0₂, 90₂, 0₂, 90₂, 0₂]



DEBONDINGS
ISOLATED/CONNECTED

[0, 90, 0, 90, 0, 0, 90, 0, 90, 0]

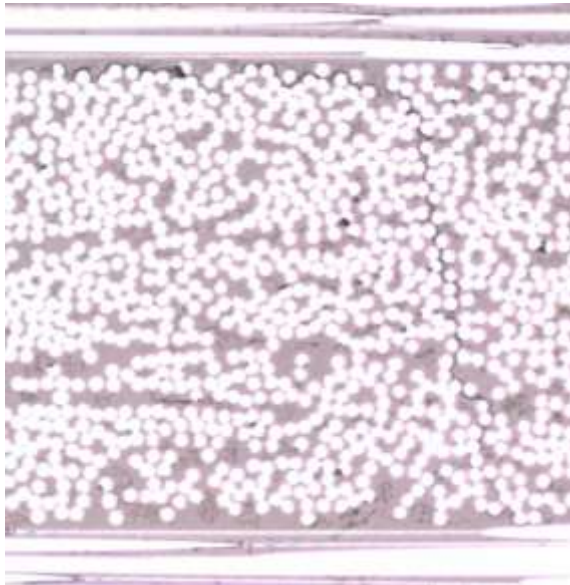


ISOLATED
DEBONDINGS

PROGRESSION OF DAMAGE IN LAMINAS ORIENTED 90°

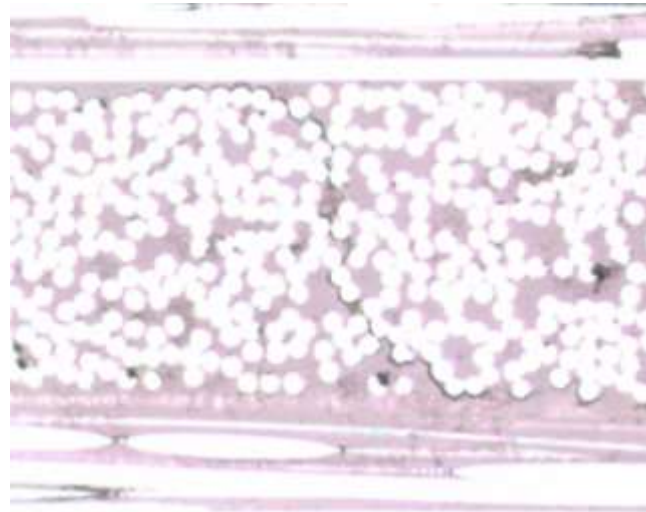
29000N-30000N
 $\varepsilon \sim 1.15\%$

$[0_3, 90_4, 0_3]$



TRANSVERSE CRACK WITH
DELAMINATION

$[0_2, 90_2, 0_2, 90_2, 0_2]$

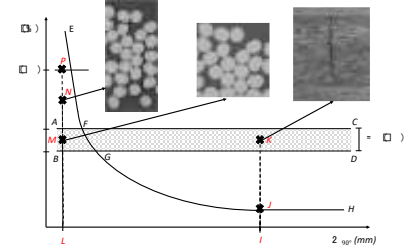
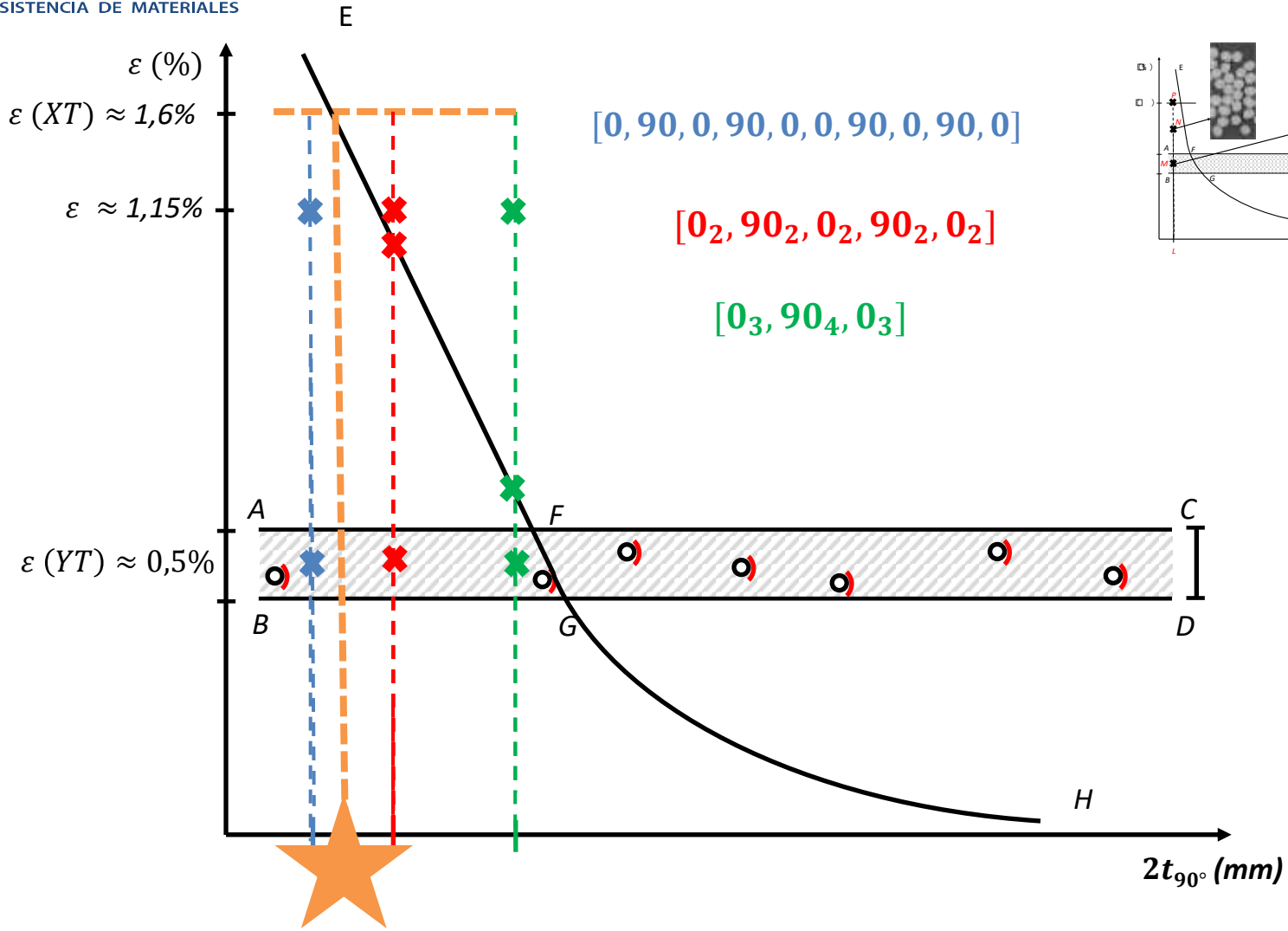


TRANSVERSE CRACK WITH
DELAMINATION

$[0, 90, 0, 90, 0, 0, 90, 0, 90, 0]$



DEBONDINGS
ISOLATED/CONNECTED

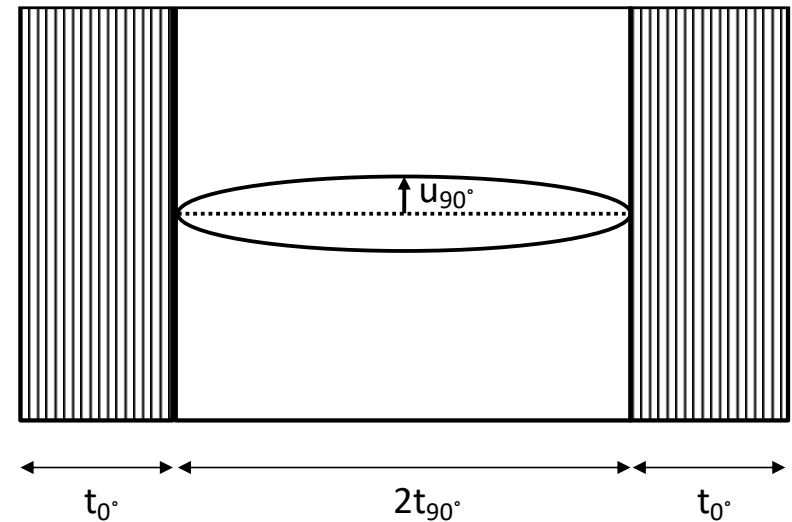
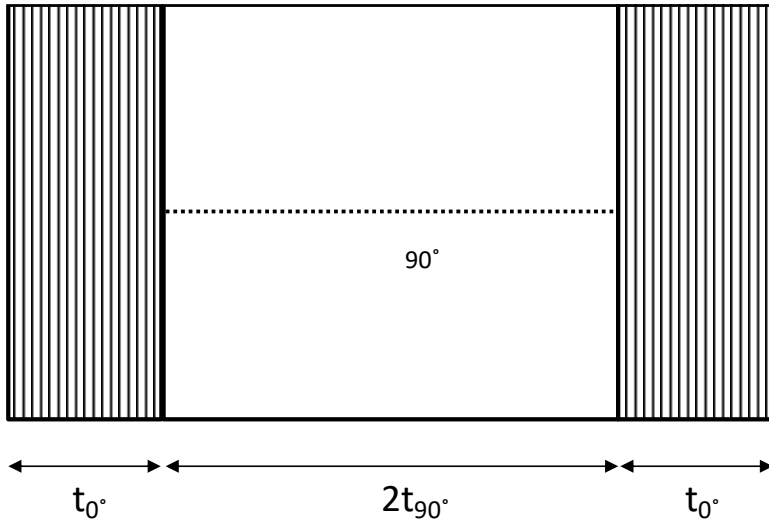


PRACTICAL APPLICATION OF THE KNOWLEDGE ACQUIRED



- Verification of the pattern of damage associated with the failure of the 90 degrees lamina as a function of the thickness
- On the concept of “in-situ Strength”
- Optimization of the thickness of the 90 degrees laminas
- **Characterization of the material: Fracture toughness of the 90 degrees lamina**

To identify Properties of AS4/8552 (thickness of one lamina 0.190 mm)

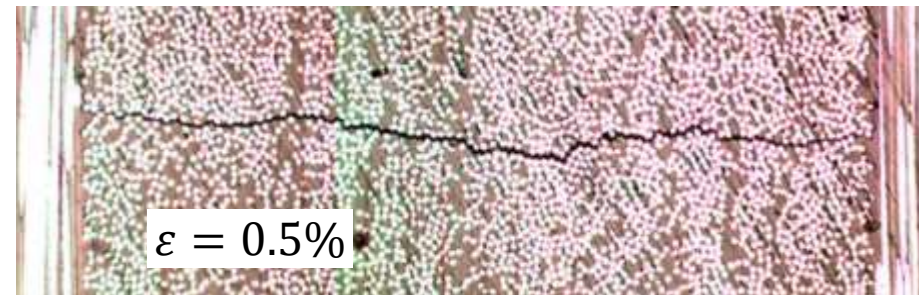


$$[0_4, 90_n, 0_4]$$

$$n = 4 (2t_{90^\circ})$$

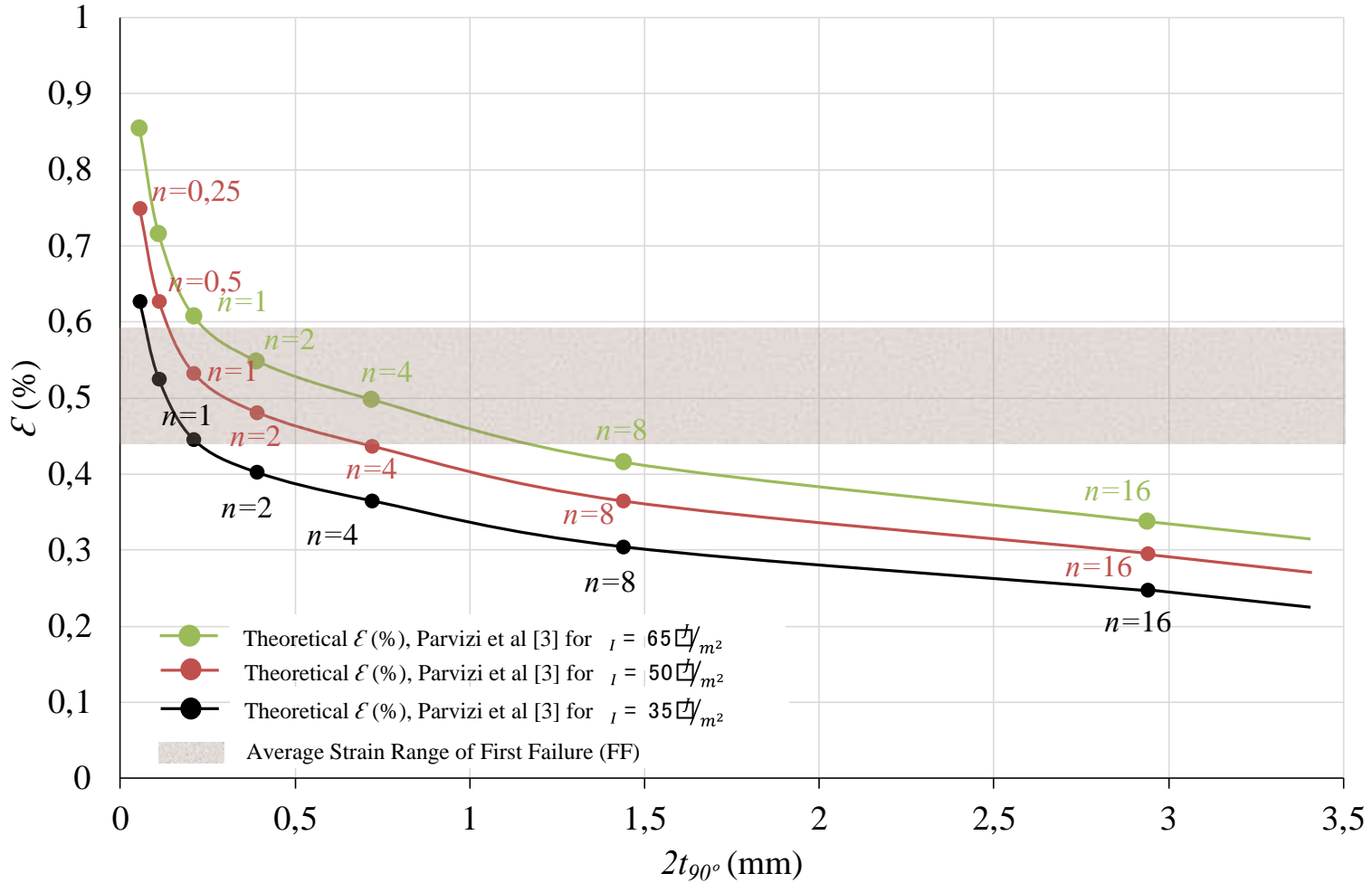
$$\int_0^{t_{90^\circ}} \left(\frac{1}{2} \cdot \sigma_{90^\circ} \cdot u_{90^\circ} \right) \cdot 2 \cdot 2 \, ds = G_{IC}^{90^\circ} \cdot 2 \cdot t_{90^\circ}$$

$$G_{IC}^{90^\circ} \approx 100 \text{ J/m}^2$$



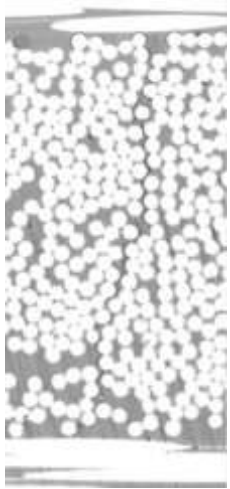
To identify Properties of AS4/8552

$[0_4, 90_n, 0_4]$

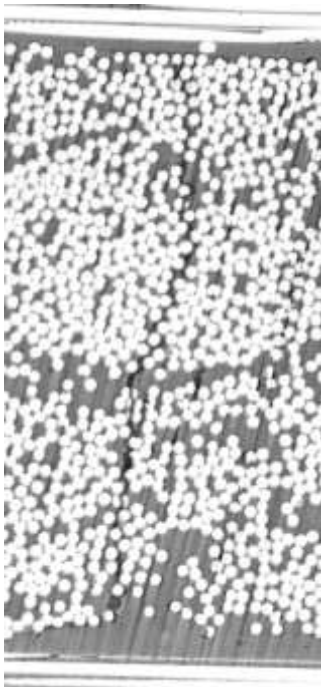


To identify Properties of AS4/8552 $[0_4, 90_n, 0_4]$

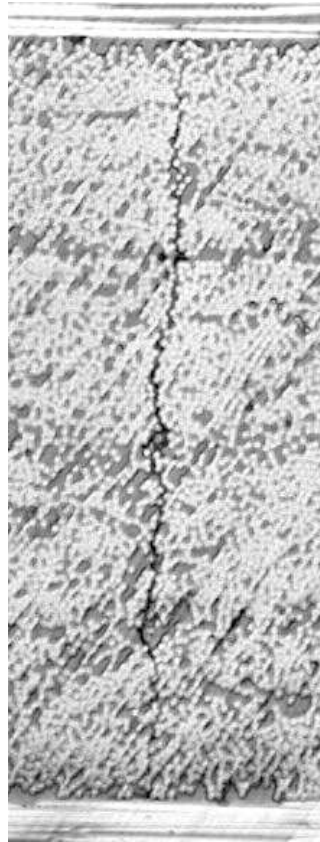
Damage patterns corresponding to the range of $\varepsilon(Y_T)$



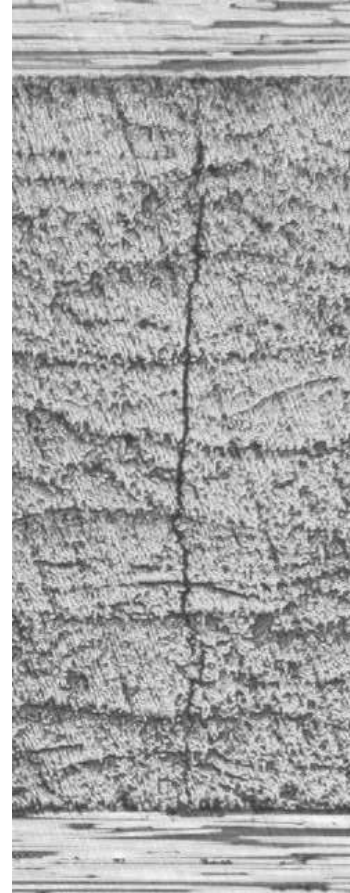
n=1



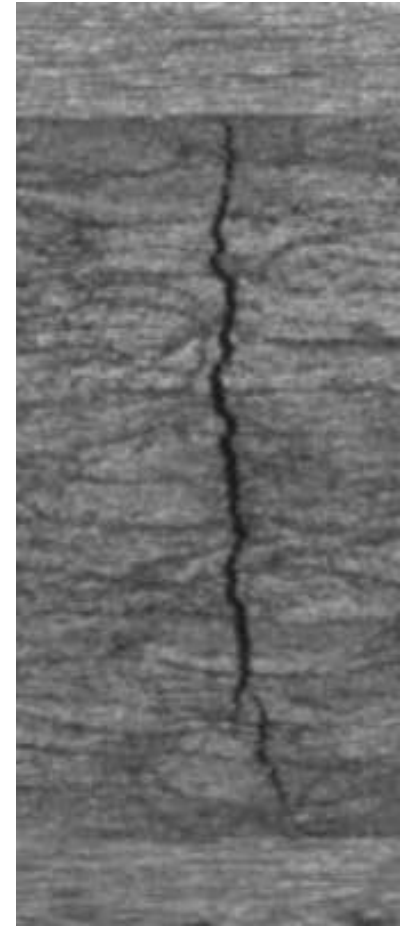
n=2



n=4

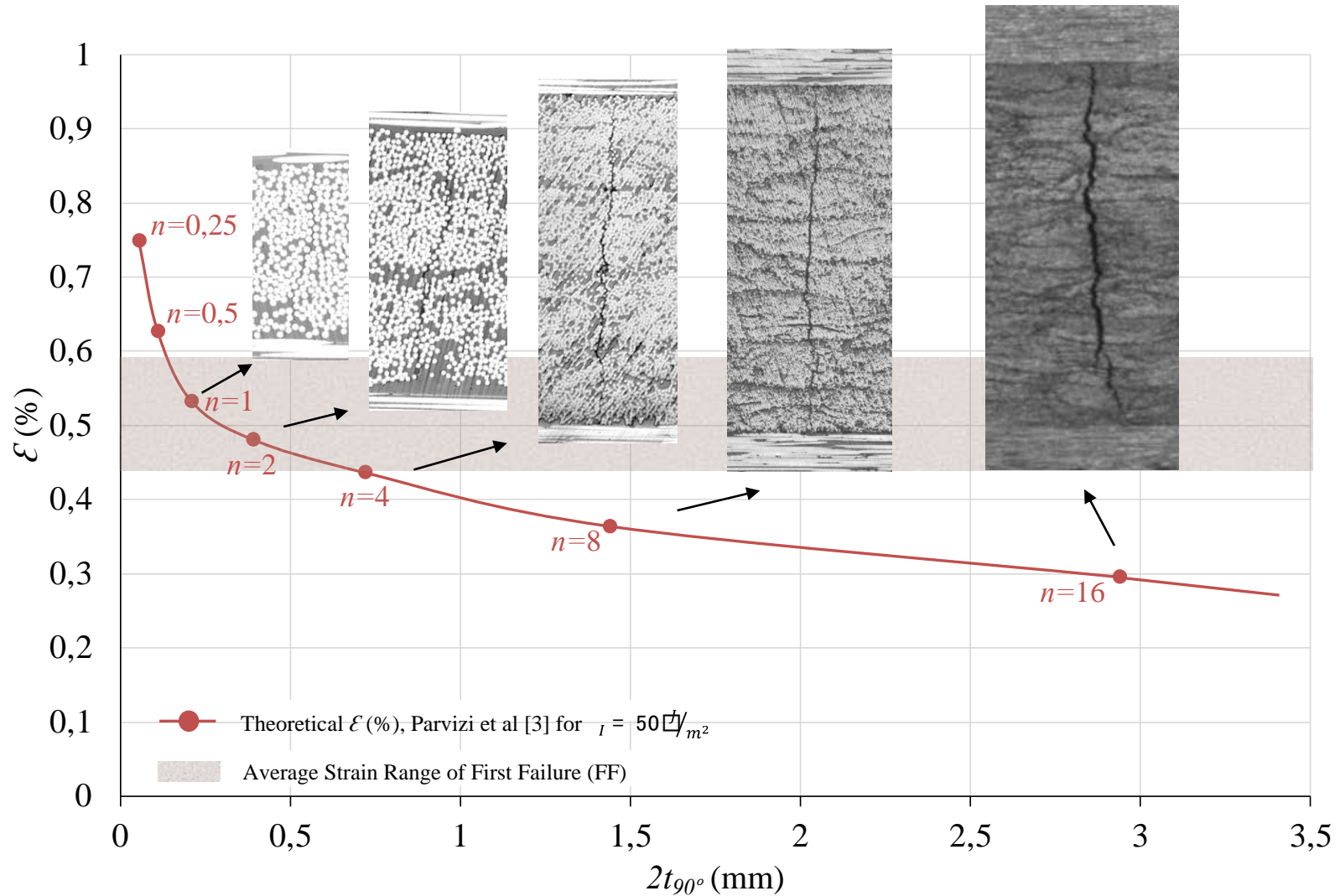


n=8



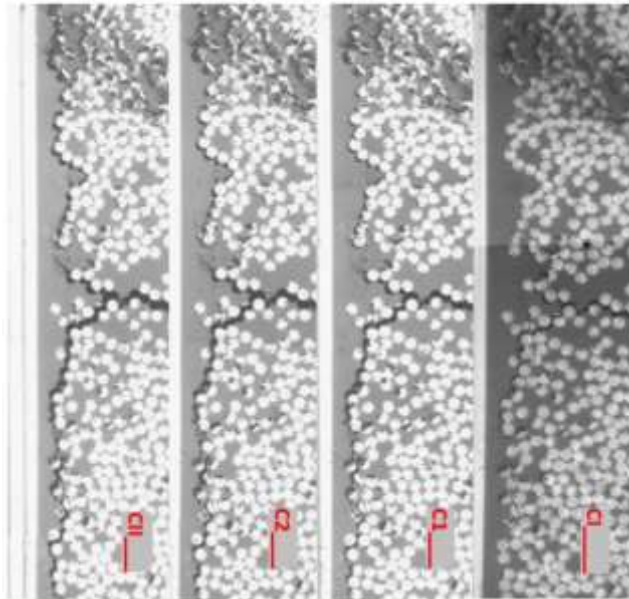
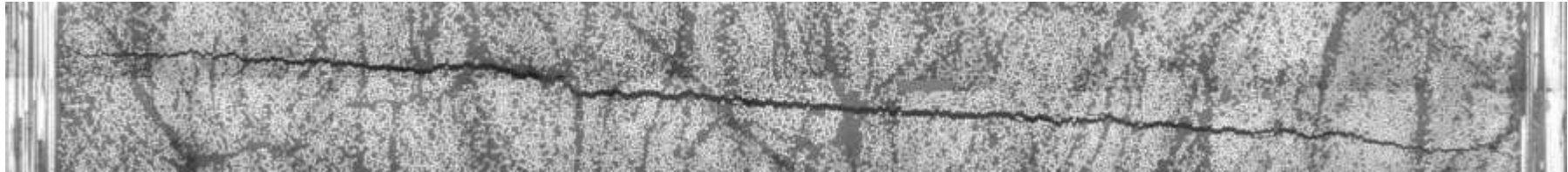
n=16

To identify Properties of AS4/8552

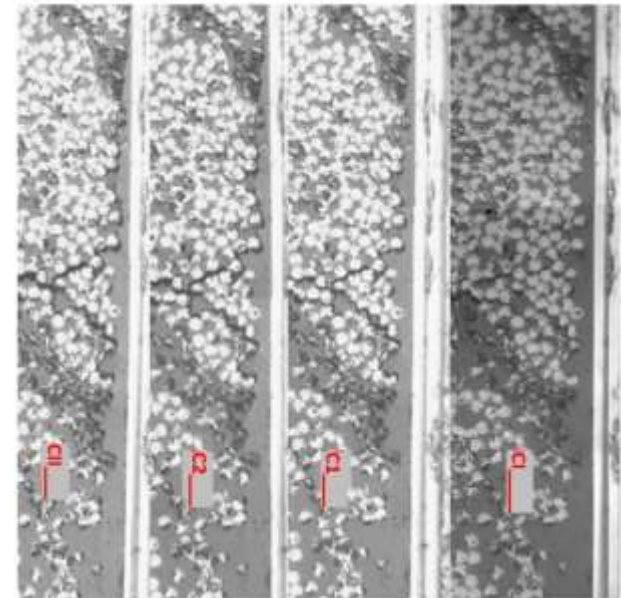


- Verification of the pattern of damage associated with the failure of a “thick” 90 degrees lamina
- On the concept of “in-situ Strength”
- Optimization of the thickness of the 90 degrees laminas
- Characterization of the material: Fracture toughness of the 90 degrees lamina
- **Future extension of the knowledge acquired**
 - Fracture toughness of the interface between 0 and 90 plies

Tests with thick plies involving significant damage between 0 and 90 degrees laminas



CI: Appearance of the first transverse crack
C1: First Strain Increase
C2: Second Strain Increase
CII: Appearance of the second transverse crack



- Verification of the pattern of damage associated with the failure of a “thick” 90 degrees lamina
- On the concept of “in-situ Strength”
- Optimization of the thickness of the 90 degrees laminas
- Characterization of the material: Fracture toughness of the 90 degrees lamina
- Present extension of the knowledge acquired
 - Fracture toughness of the interface between 0 and 90 plies
 - Effect of the different patterns of damage found with UTPs, conventional and thick plies in presence of other loadings (bending, shear, compression, fatigue,....)
 - Behaviour of a composite under biaxial loading transverse to the direction of the fibres
 - Fatigue: effect of the use of Ultra Thin Plies

- 1.- A physically based explanation, not involving Fitting Parameters, has been given about the “scale effect” in composites.
- 2.- A revision on the “in-situ strength” concept has been given.
- 3.-The explanation gives a new insight into the necessity of having a failure criterion based on a stress and an energy, as the damage is always controlled by energy, which may be different depending on the mechanism of damage activated.
- 4.- The understanding of the mechanisms of damage and failure in composites opens a discussion about the capacity of actual tools of modelling damage to represent these mechanisms. These tools have to be used once the mechanisms have been verified.
- 5.- The use of thick laminae generates more “advanced” damage in the laminate. They may unchain catastrophic failure in the laminate in presence of other loading cases (bending, torsion, fatigue,...).

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Federico París

Group of Elasticity and Strength of Materials
Department of Continuum Mechanics
School of Engineering, University of Seville, Spain

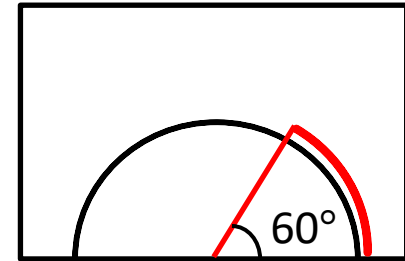
COMPTEST

Girona, May 30-June 2, 2023

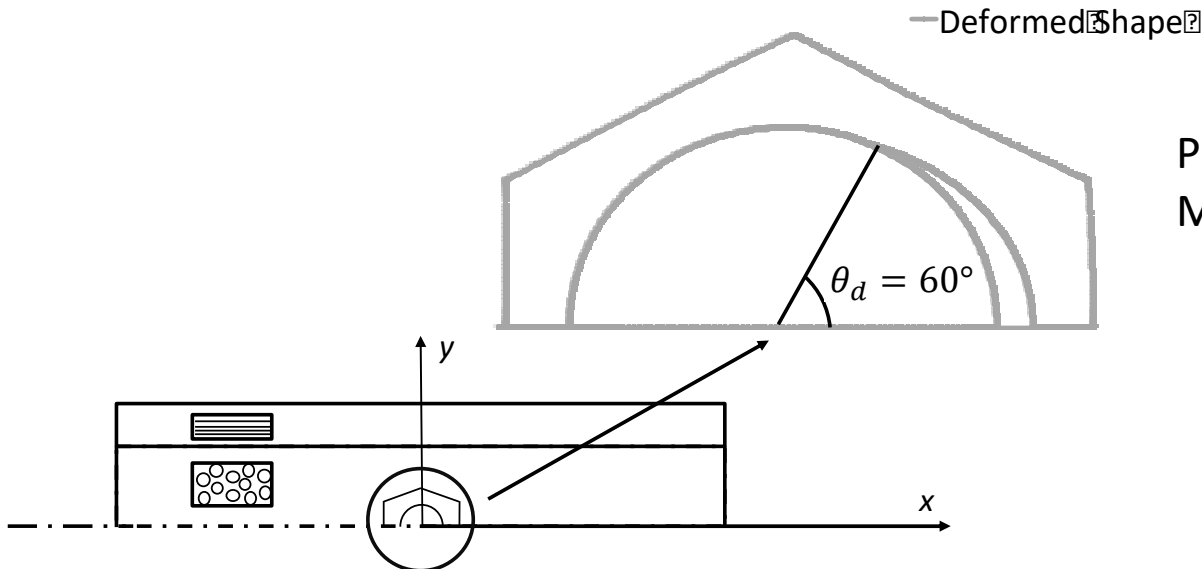
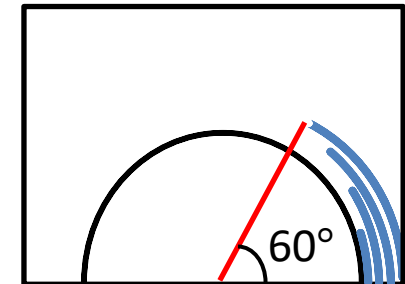
DIFFERENT WAYS OF GENERATION OF A DEBONDING

Isolated debonding between fibre and matrix which can appear in an explosive or in a progressive way.

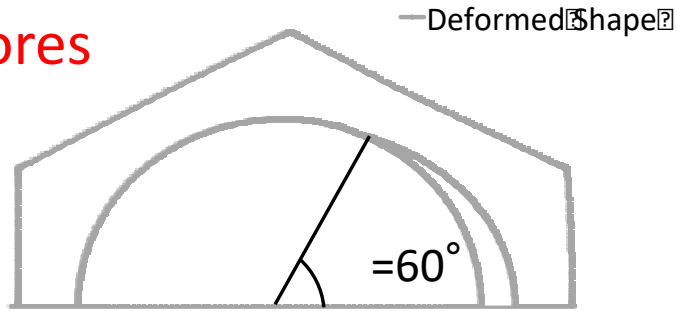
Explosive
Mechanism



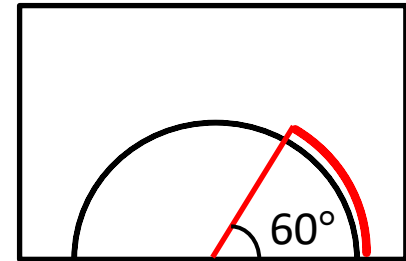
Progressive
Mechanism



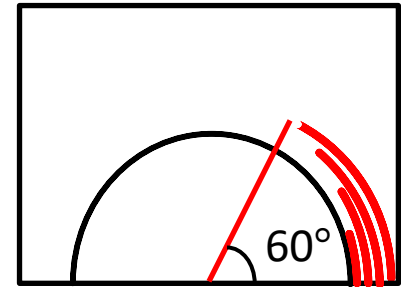
OJO con los valores de la tabla



Explosive Mechanism



Progressive Mechanism

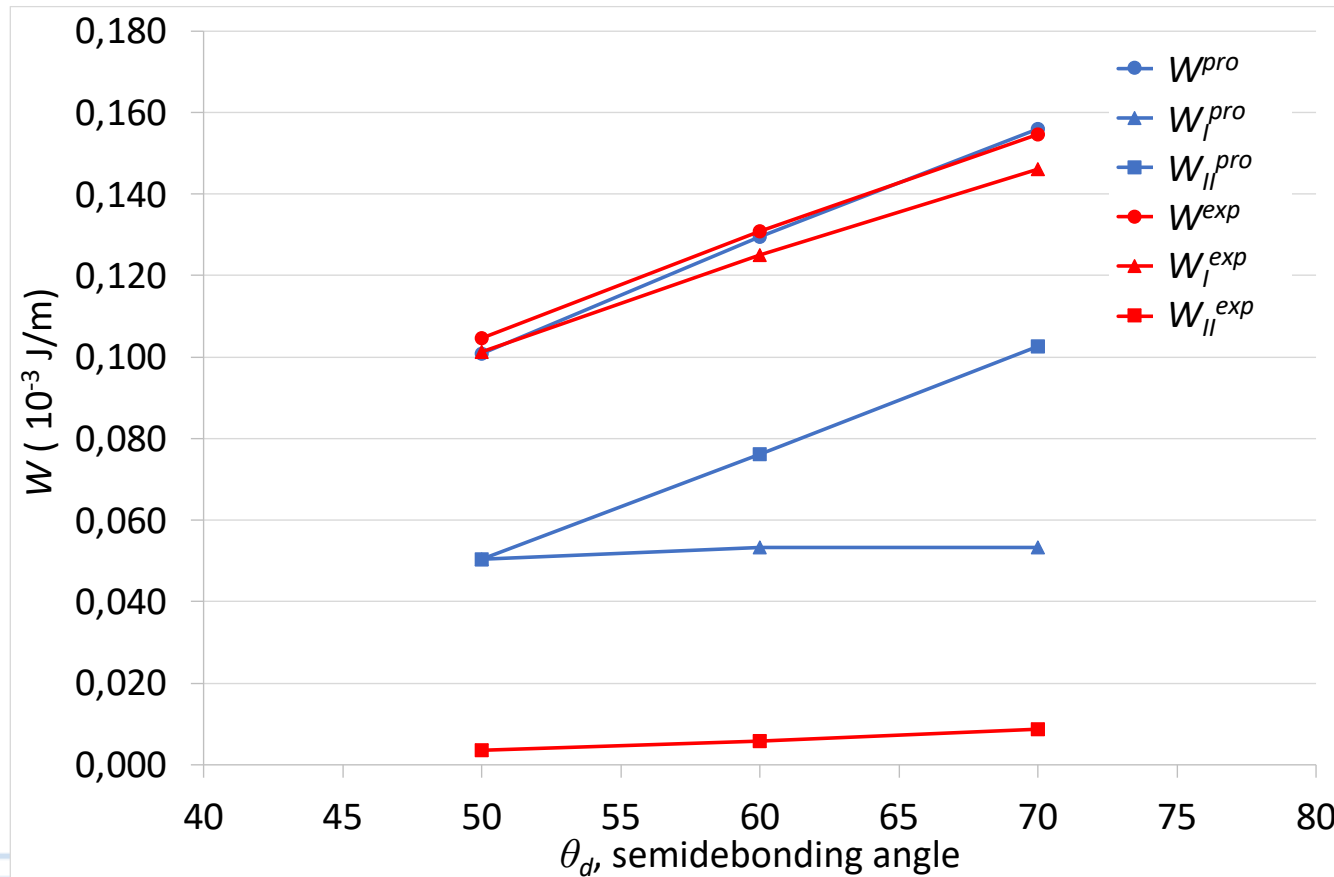


Isolated debonding between the fibre and matrix which appears in an explosive or progressive way.

If we perform in this case a micromechanical analysis with a cohesive approach we are conditioning the mechanism of damage

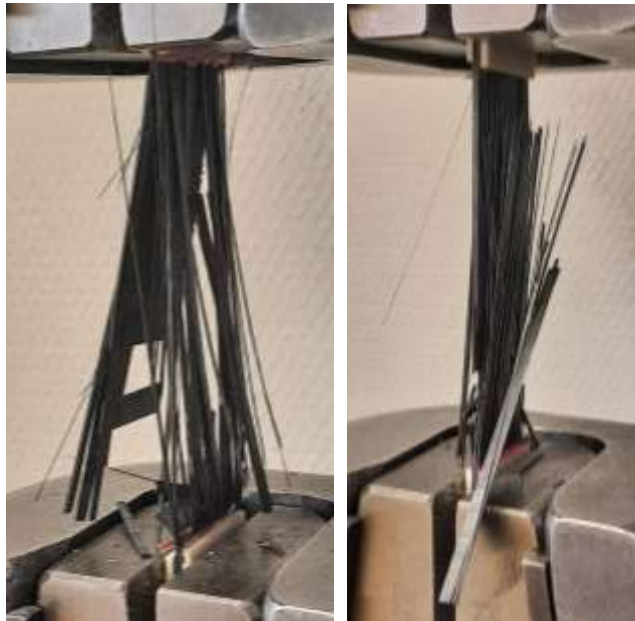
$\theta_d = 60^\circ$	PROGRESSIVE MECHANISM	EXPLOSIVE MECHANISM
$W_I (10^{-3} \text{ J/m})$	0,006667	0,031100
$W_{II} (10^{-3} \text{ J/m})$	0,009520	0,001470
$W (10^{-3} \text{ J/m})$	0,016187	0,032570

EVOLUTION OF W , W_I AND W_{II} FOR THE PROGRESSIVE AND **EXPLOSIVE** MECHANISMS OF GENERATION OF A DEBONDING

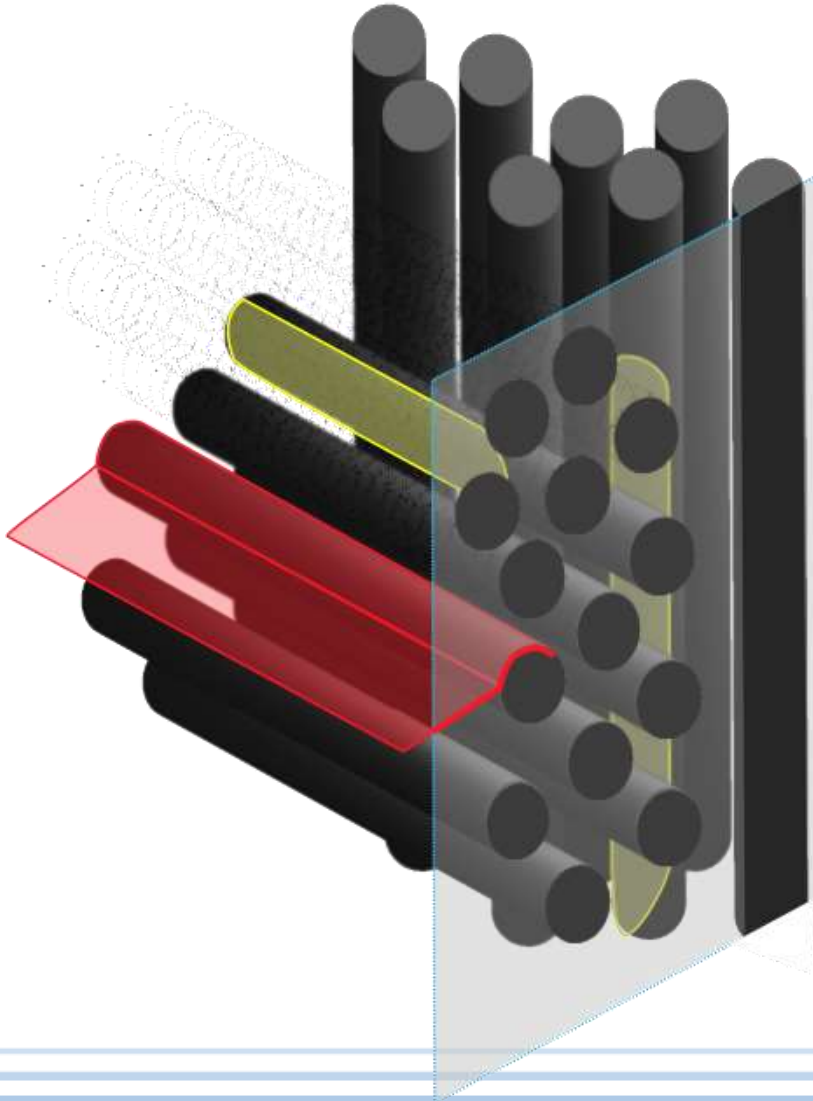


First set

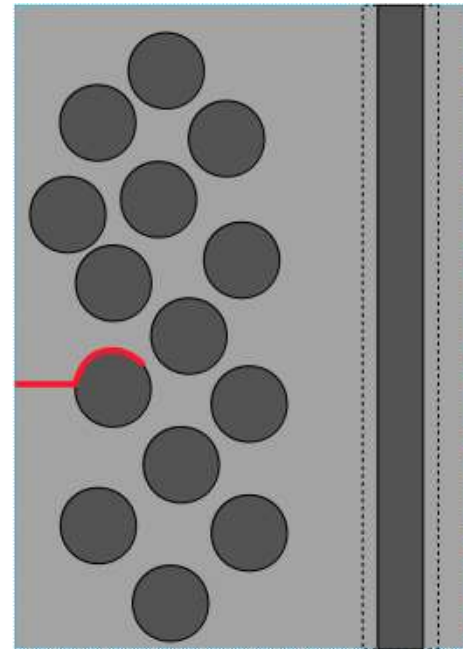
RUPTURE LOAD OF THE LAMINATES



	VALUES OF THE RUPTURE LOADS (N)		
	[0 ₃ , 90 ₄ , 0 ₃]	[0 ₂ , 90 ₂ , 0 ₂ , 90 ₂ , 0 ₂]	[0, 90, 0, 90, 0, 0, 90, 0, 90, 0]
Specimen 1	41898	56143	51235
Specimen 2	45381	54671	51748
Specimen 3	42369	49541	56230
Specimen 4	48565	54000	56829
Average	44553	53589	54010
Standard Dev.	3087	2843	2927
C.V.(%)	6,9	5,3	5,4

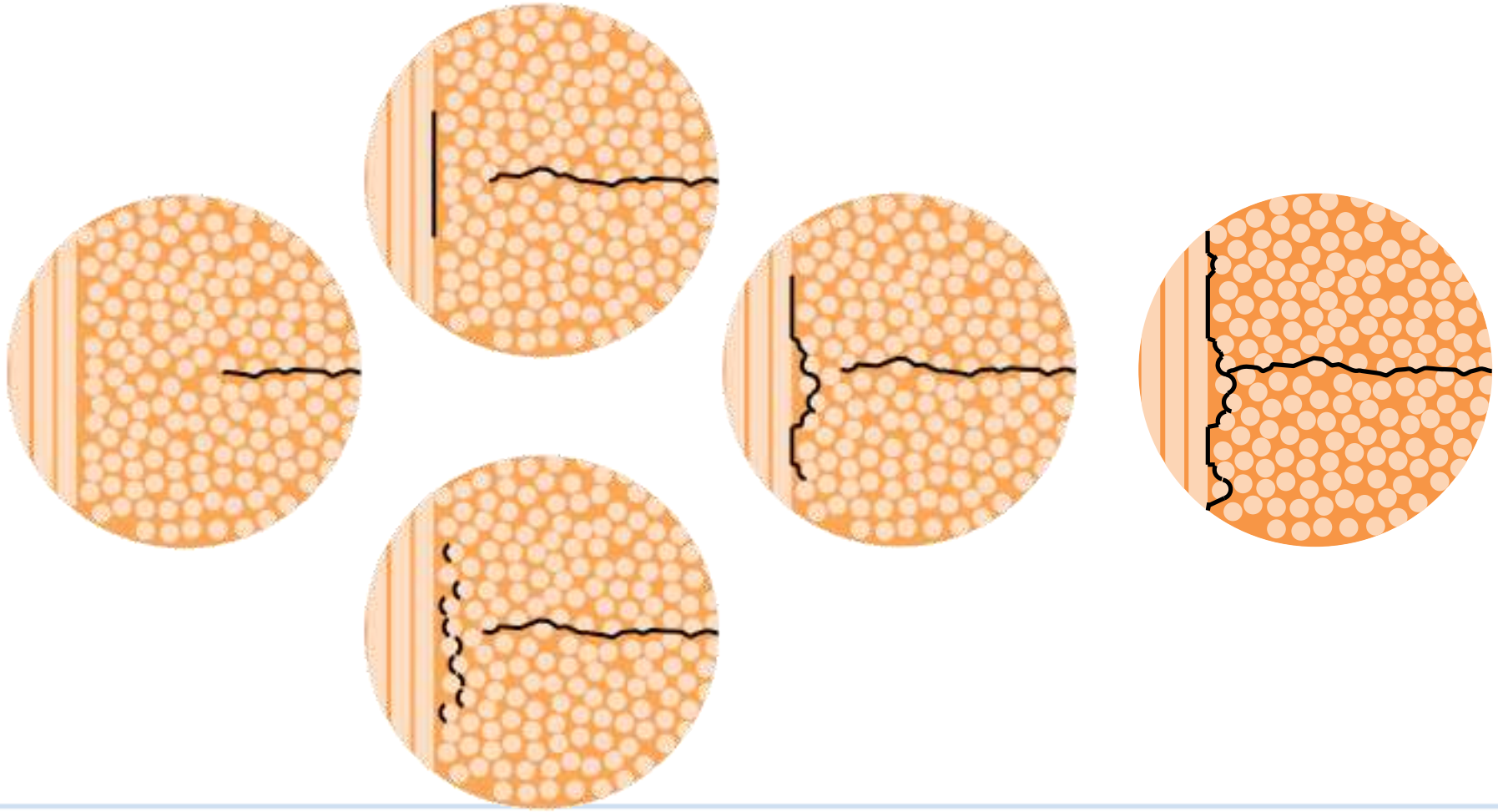


Polishing plane

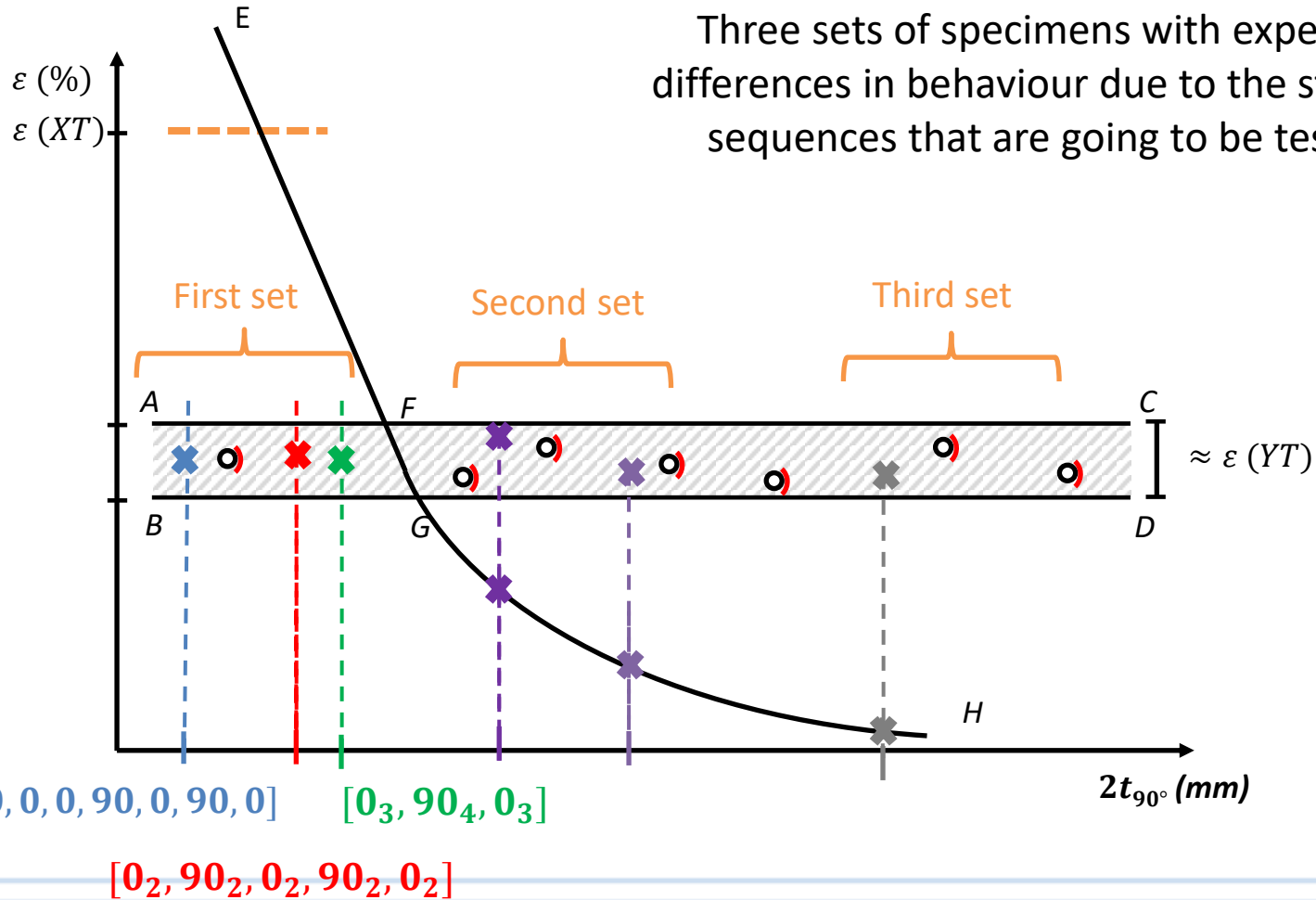


No observed damage
like delamination

2D SCHEMATIC REPRESENTATION of the GROWTH of the DAMAGE



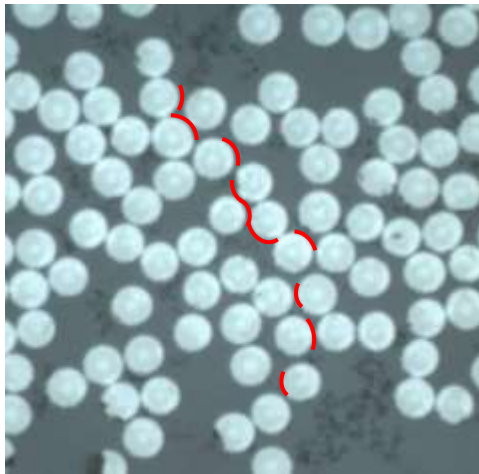
Three sets of specimens with expected differences in behaviour due to the stacking sequences that are going to be tested



DETECTION OF FIRST DAMAGE IN LAMINAS ORIENTED 90°

$[0_3, 90_4, 0_3]$

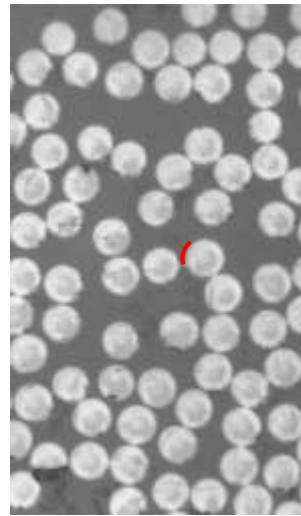
7000N-8000N



DEBONDINGS
ISOLATED/CONNECTED

$[0_2, 90_2, 0_2, 90_2, 0_2]$

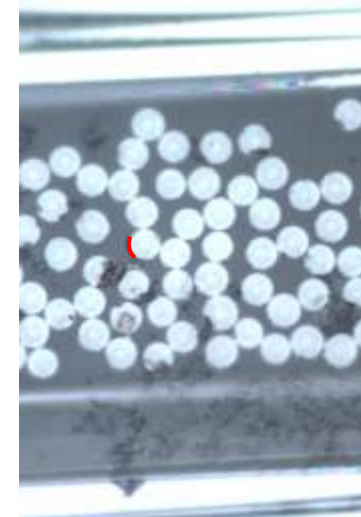
5000N-6000N



ISOLATED DEBONDINGS

$[0, 90, 0, 90, 0, 0, 90, 0, 90, 0]$

4000N-5000N

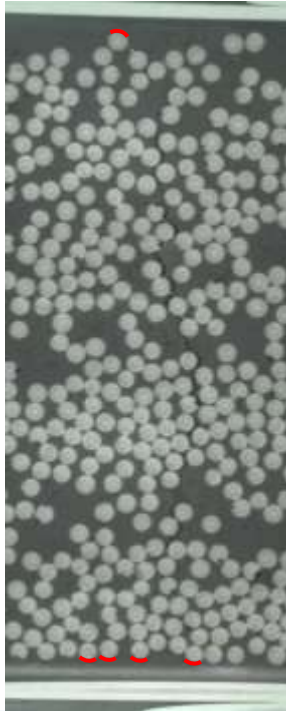


ISOLATED DEBONDINGS

PROGRESSION OF DAMAGE IN LAMINAS ORIENTED 90° IN THE FIRST SET OF SPECIMENS

20000N-21000N

$[0_3, 90_4, 0_3]$



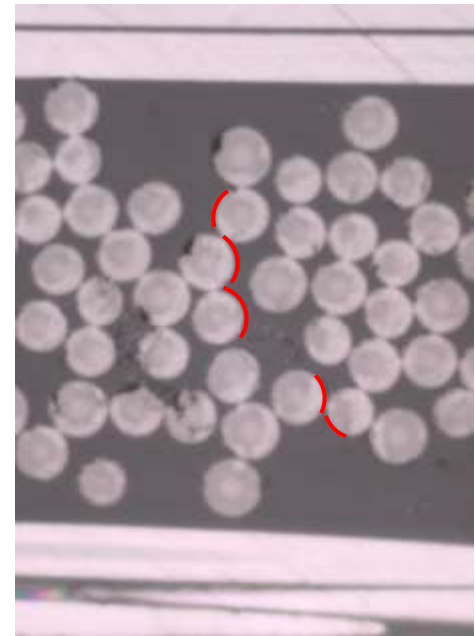
TRANSVERSE DAMAGE WITH ADDITIONAL
DAMAGE INDICATING FUTURE
DELAMINATION

$[0_2, 90_2, 0_2, 90_2, 0_2]$



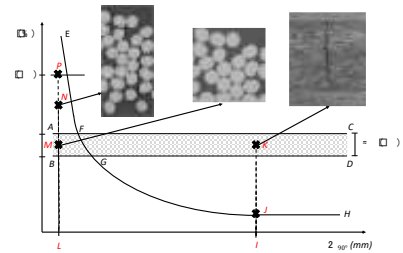
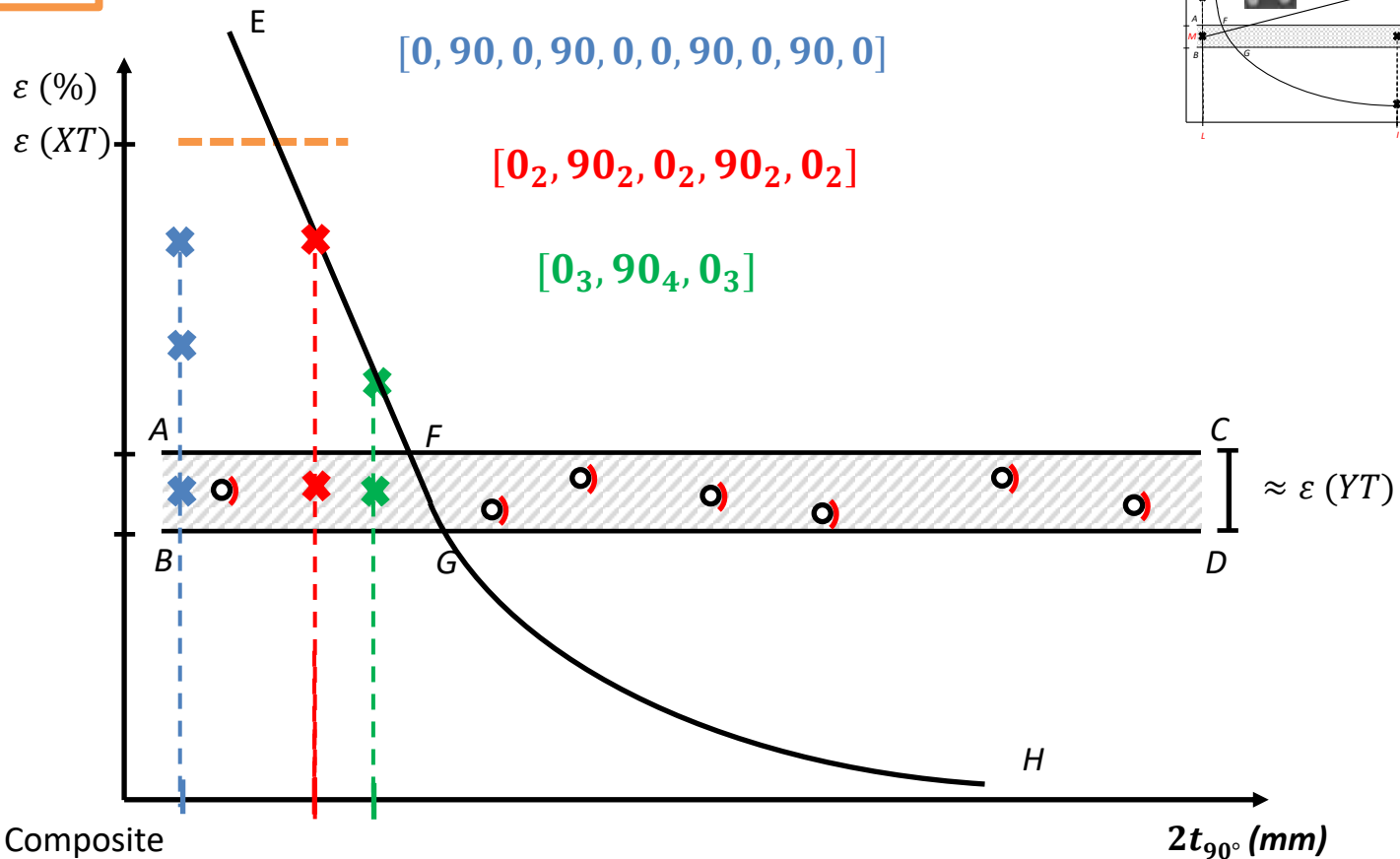
DEBONDINGS
ISOLATED/CONNECTED

$[0, 90, 0, 90, 0, 0, 90, 0, 90, 0]$



DEBONDINGS ISOLATED/CONNECTED

First set



París et al. Composite Structures, 2020.

The observations of this experimental work plenary support the explanation physically based given on the scale effect

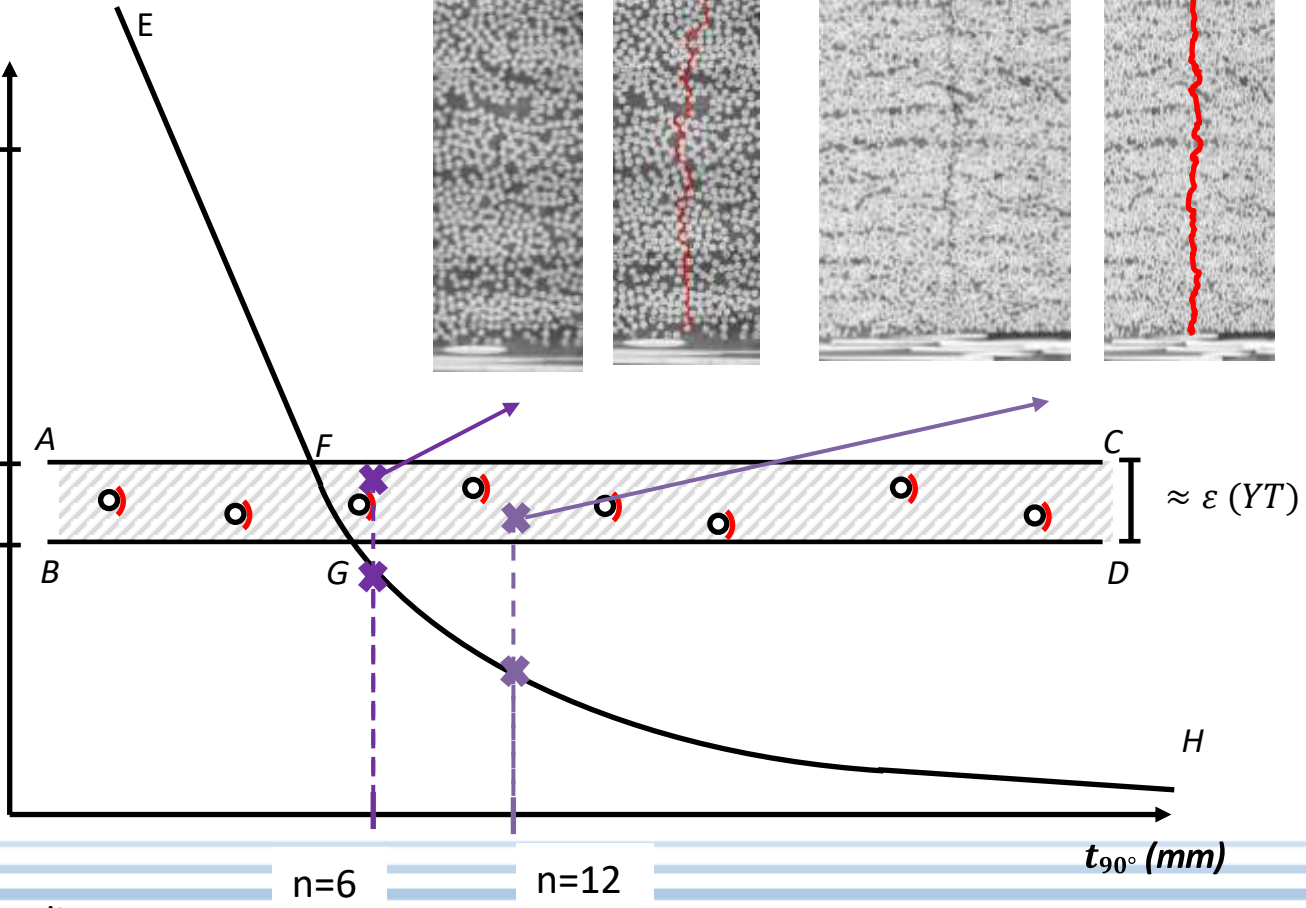
Second set with UTPs

$[0_{12}, 90_n, 0_{12}]$

n=6 and n=12

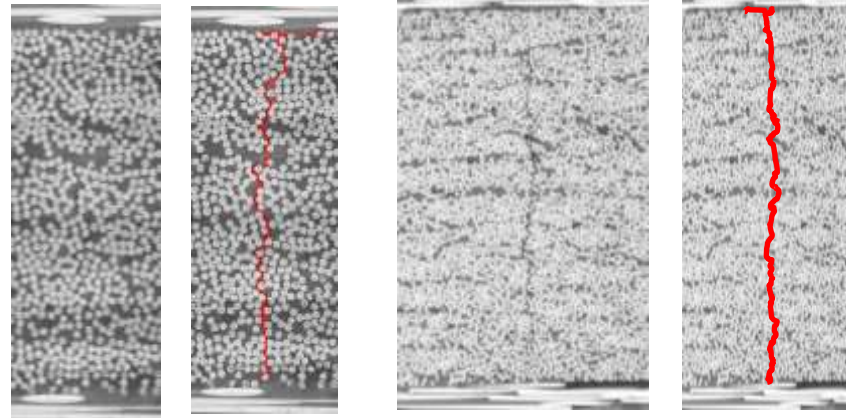
NTPT: USN50

ϵ (%)
 ϵ (XT)



300gsm

600gsm

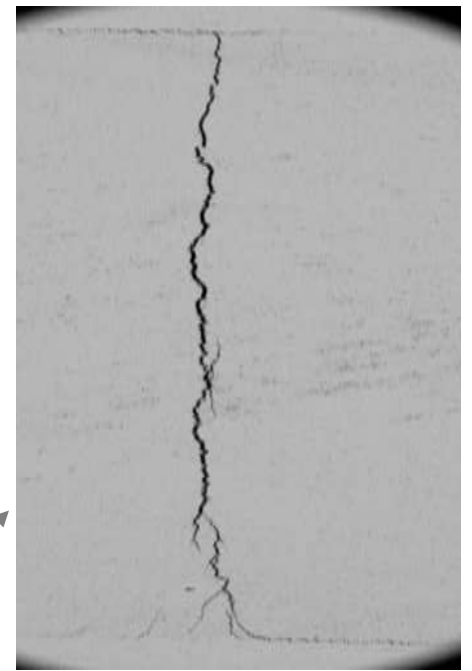
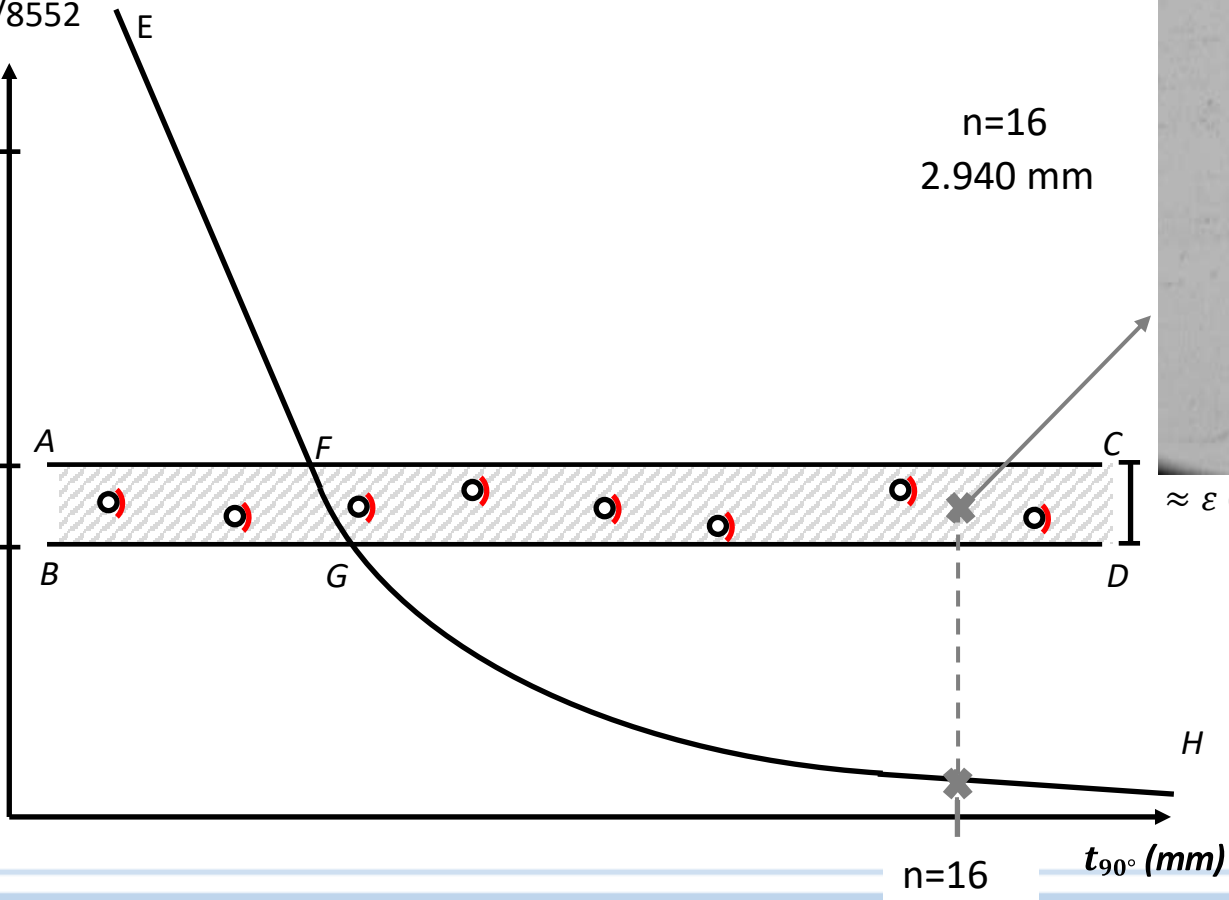


Third set

$[0_4, 90_n, 0_4]$

HEXCEL: AS4/8552

ε (%)
 ε (XT)



The purpose of this presentation is to illustrate:

- 1.-How Fracture Mechanics and Interfacial Fracture Mechanics may help engineers to understand Mechanisms of Damage of composite and its consequences in the performance of a composite
- 2.- The capacity of numerical tools to represent actual mechanisms of damage

3rd question: Why do we need a double criteria, one based on energy and another based on stress?

There is no need of having two failure criteria one based on energy and another based on stress. There are two mechanisms of failure controlled by energy and we need two criteria, one of them being easily expressed by means of a stress

The similarity with other damage scenarios is clear: the start of plastic behaviour is controlled by energy (distortion energy) but there is a very convenient stress (shear octahedric stress or even the modulus of the deviator stress tensor) that represents exactly the same

4th question: Is it adequate the in-situ strength concept?

The full understanding of the damage process explaining the Scale effect has meant a reconciliation with the in-situ strength concept, leading to questioning the concept of Strength in itself